

UV-curable photosensitive silicone resins based on a novel polymerizable photoinitiator and GO-modified TiO₂ nanoparticles

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ABSTRACT

A novel polymerizable benzophenone photoinitiator containing acryloxy (BPA) was synthesized. The photopolymerization of photosensitive methacrylic-silicone resins (MASR) initiated by BPA was studied. Results show that BPA is an efficient photoinitiator, which could generate radicals from the photolysis reaction and be consumed via copolymerizing with MASR. The designed UV-curing silicone resins system initiated by BPA showed a high carbon-carbon double bonds conversion above 80% after 20 s UV radiation. In addition, reduced graphene oxide (rGO)-TiO₂ composites were elaborated by hydrothermal reaction. The synergistic effects from graphene and TiO₂ further enhanced the carbon-carbon double bonds conversion of MASR to 86% with 15 s UV irradiation, demonstrating that the incorporated rGO-TiO₂ significantly improve the formulation UV-curing reactivity. Additionally, UV-cured MASR with rGO-TiO₂ showed an excellent thermal stability. This work proposes an efficient strategy to improve the polymerization degree of UV-curable photosensitive silicone resins.

1. Introduction

Silicone resins, as an interesting class of non-carbonic polymers, provide remarkable properties including excellent dielectric properties, good thermal stability, impressive antiweatherability and chemical resistance [1–3]. The crosslinking chemistries of silicone-based materials can be categorized into three main chemical reactions: (1) hydrosilylation between vinyls and hydride-terminated siloxanes in the presence of the metal catalyst such as platinum [4], (2) thiol-ene click chemistry between vinyl group and thiol group [5], (3) condensation reaction of alkoxy silane [6]. However, these crosslinking chemistries still present pitfalls that prevent their use for specific applications and have many drawbacks such as the harmful of heavy metal catalysts to the environment, high curing temperature and high energy consumption. Therefore, most researches focus on UV-curable silicone resins and most UV-curable system involves reactive acrylate, methacrylate, or allyl groups [7,8].

The light induced polymerization technique have been considered as ecofriendly due to its outstanding features such as high efficiency, low energy consumption, solvent-free formulation, and room temperature treatment [9]. These characteristics promote photopolymerization over thermally induced processes which require much time and high temperature to complete the reaction. Significantly, photopolymerization materials have wide application in elaboration of

coatings, adhesives, and other products [10–12].

As a crucial component of a UV-curable system, photoinitiators could absorb light and generate active radicals to initiate the polymerization [13]. However, low-molecular-weight photoinitiators are generally treated as a problem in photopolymerization due to the disadvantages such as odor, yellowing, poor compatibility with the UV-curable resin and migration in the post-cured process [14]. Thus, polymerizable photoinitiators have been developed as a valuable alternative to overcome the disadvantages, which can not only generate free radicals to initiate photopolymerization, but also copolymerize with the polymer [15].

TiO₂ has also been testified to be free-radical photoinitiators for curing (meth)acrylic resins, due to its ability of generating electron-hole pairs under photoexcitation, which could react with resin monomers providing reactive radicals [7]. Due to the unique synergistic effect, incorporation of the graphene and TiO₂ is considered a promising nanocomposite [16]. Graphene, which is often prepared by reducing graphene oxide, have drawn interest from researchers owing to its large specific surface area, excellent electrical conductivity, high charge carriers mobility and high mechanical strength [17–19]. It turns out to be an electron acceptor/transporter for TiO₂ particles and anticipates to significantly enhancing the lifetime of electron-hole pairs [20]. Thus, it is a potential candidate to facilitate the adsorption capability and enhance the photoactivity of TiO₂ under UV light due to the unique

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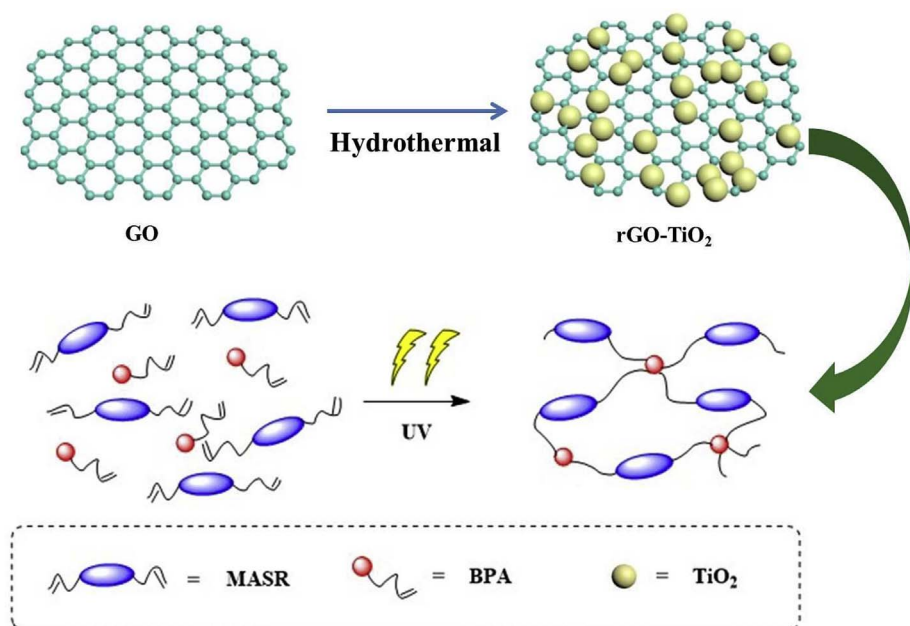


Fig. 1. Schematic of the preparation procedure of UV-cured MASR.

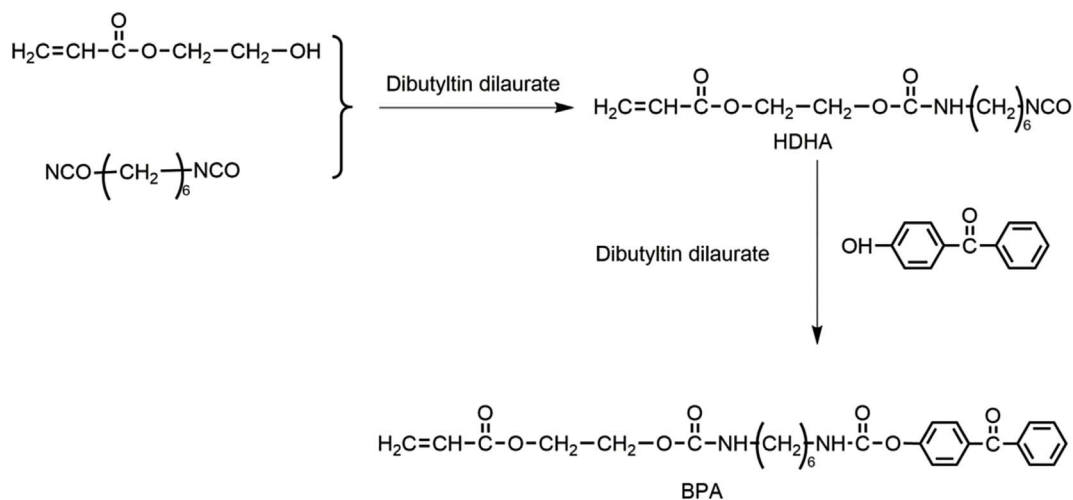


Fig. 2. The synthesis routes of BPA.

electron-transferring property. In addition, the incorporation of graphene could avoid TiO₂ particles suffering aggregation in an organic matrix, which consequently leads to good dispersity.

In this study, polymerizable photoinitiator was synthesized to initiate the polymerization of UV-curable photosensitive methacrylic-silicone resins. The effects of BPA and rGO-TiO₂ composites on the photopolymerization of MASR were investigated. The photoinitiation mechanism and the photopolymerization behavior were also presented. The schematic representation of UV-cured methacrylic-silicone resins based on a BPA and rGO-TiO₂ was shown in Fig. 1.

2. Experimental

2.1. Materials

Methyltriethoxysilane (MTES), dimethyldiethoxysilane (DDS), 3-methacryloxypropyltrimethoxysilane (MPTS) were obtained from Qufu Chenguang Chemical Co., Ltd. (China). Hexamethylene diisocyanate (HDI), 4-hydroxybenzophenone, 2-hydroxyethyl acrylate (2-HEA), dibutyltin dilaurate, dibutylamine and 2-(dimethylamino)ethyl methacrylate (DMAEMA) were supplied by Aladdin Reagent Shanghai Co.,

Ltd (China). Tetrabutyl titanate (Ti(OC₄H₉)₄), potassium permanganate, and concentrated sulfuric acid (98 wt%) were obtained from Sinopharm Chemical Reagent Co., Ltd (China). All chemical reagents were used as received.

2.2. UV-Curing of MASR based on BPA and GO- TiO₂

MASR with initiator (0.5 μm film on glass) was irradiated by a light source of high-pressure mercury lamp (1000 W) with a distance of 10 cm from lamp to specimen at room temperature in air atmosphere. The composition of the samples investigated in the work was shown in Table S2. DMAEMA as the coinitiator, [BPA]/[DMAEMA] = 1:4 mol/mol [21].

2.2.1. Preparation of MASR

MTES, DDS and MPTS with different mole ratio 7:2:1 and 6:3:1 (supported in Table S1) were added into a flask connecting to mechanical stirrer and refluxing condenser. Then hydrochloric acid solution (10 wt%, diluted by ultrapure water) was slowly added dropwise through the constant-pressure funnel to adjust the pH value to 3–4. The blend was then refluxed at 60 °C for 4 h. The transparent viscous

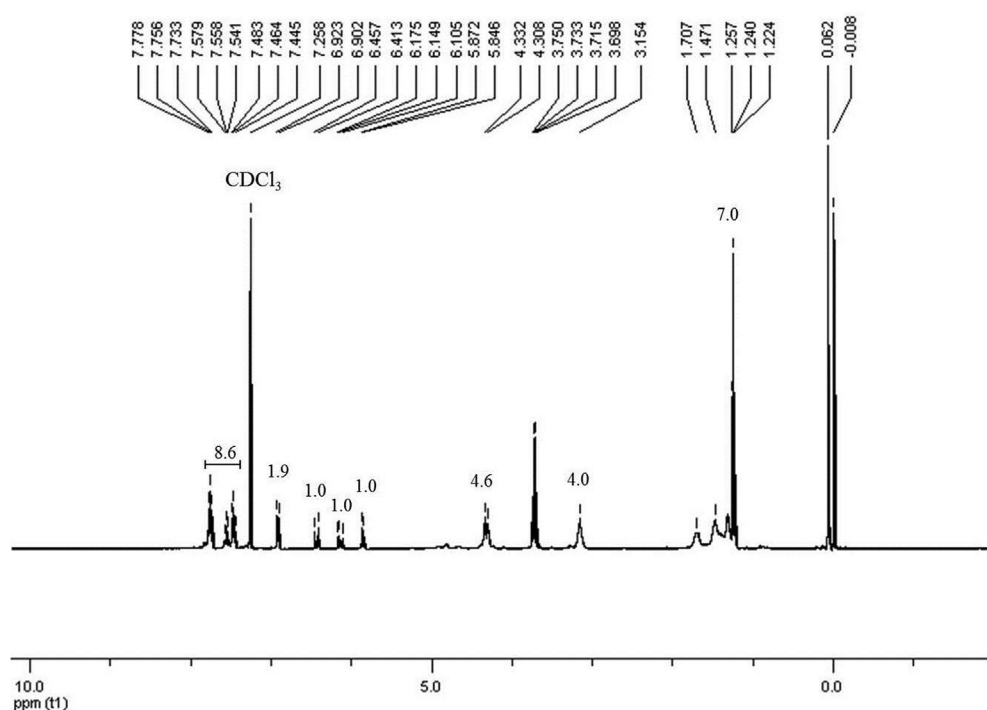
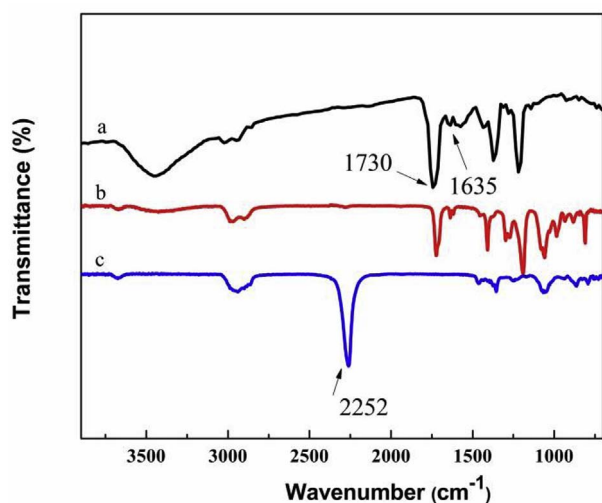
Fig. 3. ^1H NMR spectrum of BPA.

Fig. 4. FTIR spectra of a) BPA; b) 2-HEA; c) HDI.

product was obtained after dried in rotary evaporator at 50 °C for 1 h. The characterization data of MASR was provided in supporting information.

2.2.2. Preparation of BPA

In a three-necked 500 mL flask equipped with a magnetic stirrer, a reflux condenser, a thermometer and a N_2 gas inlet, a solution of HDI (2.35 g, 14 mmol) in DMF was added. The temperature was then raised to 50 °C and a solution of 2-HEA (1.63 g, 14 mmol) in DMF containing dibutyltin dilaurate (0.5 wt%) was added dropwise into the flask. The reaction mixture was maintained at 50 °C for 8 h till the concentration of NCO reached the theoretical value (the percentage of residual NCO groups was determined by typical dibutylamine chemical titration). Then, the solution of 4-hydroxybenzophenone (2.78 g, 14 mmol) in DMF was added dropwise over a period of 1 h and reacted with the residual NCO groups at 65 °C for 15 h with adding extra dibutyltin dilaurate (1.0 wt%). The precipitated product was isolated by suction

filtration and dried in vacuum oven.

2.2.3. Preparation rGO-TiO₂ composites

Graphene oxide (GO) was prepared by a modification of Hummers' method [22]. rGO-TiO₂ composites were synthesized hydrothermally. Briefly, the volume ratio of ethanol/water was 75:5 in this reaction system. The weight ratios of raw material $\text{Ti}(\text{OC}_4\text{H}_9)_4/\text{GO}$ were 400:1, 200:1, 100:1 and 50:1, which were denoted as rGO-TiO₂-1, rGO-TiO₂-2, rGO-TiO₂-3, and rGO-TiO₂-4, respectively. The reactants was maintained and stirred at 80 °C for 12 h and then poured into a Teflon-lined autoclave for hydrothermal reaction at 200 °C for 5 h. Finally, rGO-TiO₂ composites were collected after washed repeatedly with ethanol and deionized water and dried. Then different rGO-TiO₂ percentage of the resins by weight (1%, 2%, 5%, 10%) was added into silicone resin to evaluate their influence on UV-curing reactivity and thermal stability.

2.3. Characterizations

Fourier transform infrared (FTIR) spectra in the range of 4000–400 cm^{-1} were obtained by a Nicolet FTIR 5700 spectrometer (USA). The molecular weight of the obtained methacrylic-silicone resins was characterized by gel permeation chromatography (GPC, Water 515, Waters, USA), using THF as an eluent and polystyrene as the standard. ^1H NMR and ^{29}Si NMR spectra were provided from a Bruker WM500 (Germany). UV spectra were analyzed utilizing a UV-vis spectrophotometer (Model Lambda 900, USA). The X-ray diffraction (XRD) data were acquired from a Shimadzu X-ray diffractometer. The X-ray photoelectron spectroscopy (XPS) was analyzed with a VG electron spectrometer having a 0.3–0.5 eV resolution from a monochromated aluminum anode X-ray source using $\text{K}\alpha$ radiation with 1486.6 eV (ESCALAB Mk II, UK). The Raman spectra of the samples were performed using a laser confocal Raman spectrometer (Renishaw inVia, England). The thermal stability of the UV-cured samples were studied on a NETZSCH thermoanalyzer (STA449C, made in Germany) under a nitrogen atmosphere with the heating ramp of 20 °C/min from 30 to 900 °C.

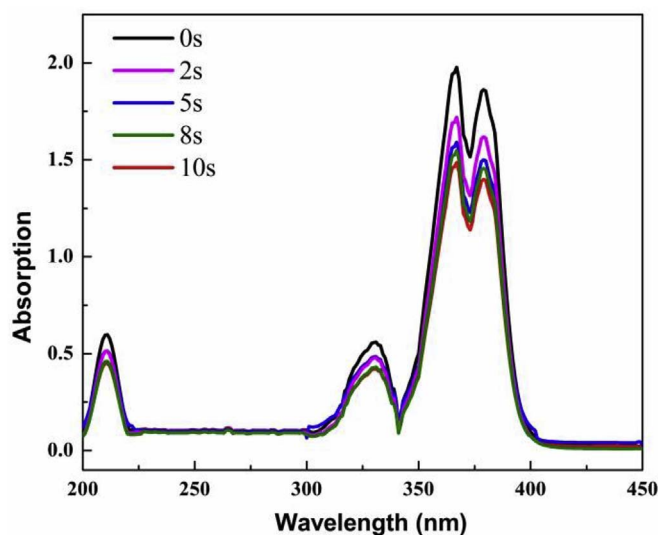


Fig. 5. UV absorption spectrum of BPA (1 wt%) in MASR solution with different UV radiation time.

3. Results and discussion

3.1. Synthesis and structural characterization of BPA

In order to enhance UV-curing reactivity and the conversion of the C=C double bonds of MASR, polymerizable photoinitiator containing acryloxy was prepared (see Fig. 2). To avoid the side reaction, the intermediate (HDHA) was firstly synthesized by HDI and 2-HEA with catalyst dibutyltin dilaurate, then benzophenone was incorporated into HDHA in order to obtain the BPA. The molecular structure of the target product was confirmed by ^1H NMR and FTIR (Fig. 3 and Fig. 4).

The structure of BPA was characterized by ^1H NMR and FTIR. In ^1H

NMR spectrum (Fig. 3), the signals at 1.224–1.257 ppm, 3.154 ppm and 4.308–4.332 ppm are assigned to the methylene protons, indicating the existence of methylene group derived from the HDI and 2-HEA. The chemical shift at 6.902–6.923 ppm is attributed to amino protons and the aromatic protons are detected at the range of 7.445–7.778 ppm. The protons of $-\text{HC}=\text{CH}_2$ appear at 5.846–6.457 ppm. The characteristic ^1H NMR data of the synthesized compound was given in Table S3.

The expected structure of BPA was further evidenced by FTIR as depicted in Fig. 4. For BPA, the characteristic peak at 1730 cm^{-1} assigned to the stretching vibration of C=O can be observed and the absorption band located at 1635 cm^{-1} is ascribed to C=C. The feature at 3444 cm^{-1} refers to the stretching vibration of secondary amine group $-\text{NH}-$. The peaks at $2800\text{--}2900\text{ cm}^{-1}$ indicate that $-\text{CH}_2-$ are existed in the target products. Besides, the characteristic stretching vibration of $-\text{NCO}$ group completely disappears in FTIR spectrum, indicating the successful synthesis of BPA.

UV absorption spectra are utilized to analyze the absorption property of BPA (see Fig. 5). The maximum of absorption of BP is around 250 nm, which can be attributed to the main benzenoid $\pi-\pi^*$ type transition [23]. However, BPA exhibits a markedly red-shifted maximal absorption, which may be ascribed to the strong electron-donation effect via the influence of substitutional group [21]. It can be found that the higher maximal absorption is beyond 300 nm, which is much higher than those polymerizable photoinitiators [24], and is similar to the thioxanthone photoinitiator [25]. It is well known that the maximal absorption of photoinitiators is very important in terms of their photochemical activity. Thus, the red-shifted maximum makes BPA attractive as a photoinitiator, suitable for UV-curing near the visible spectrum. In addition, the absorption intensity decreases as irradiation time increases. At the end of the irradiation, the higher maximal absorbance reduces from 1.979 to 1.485, suggesting that the photoinitiators are consumed as the photolysis proceeds.

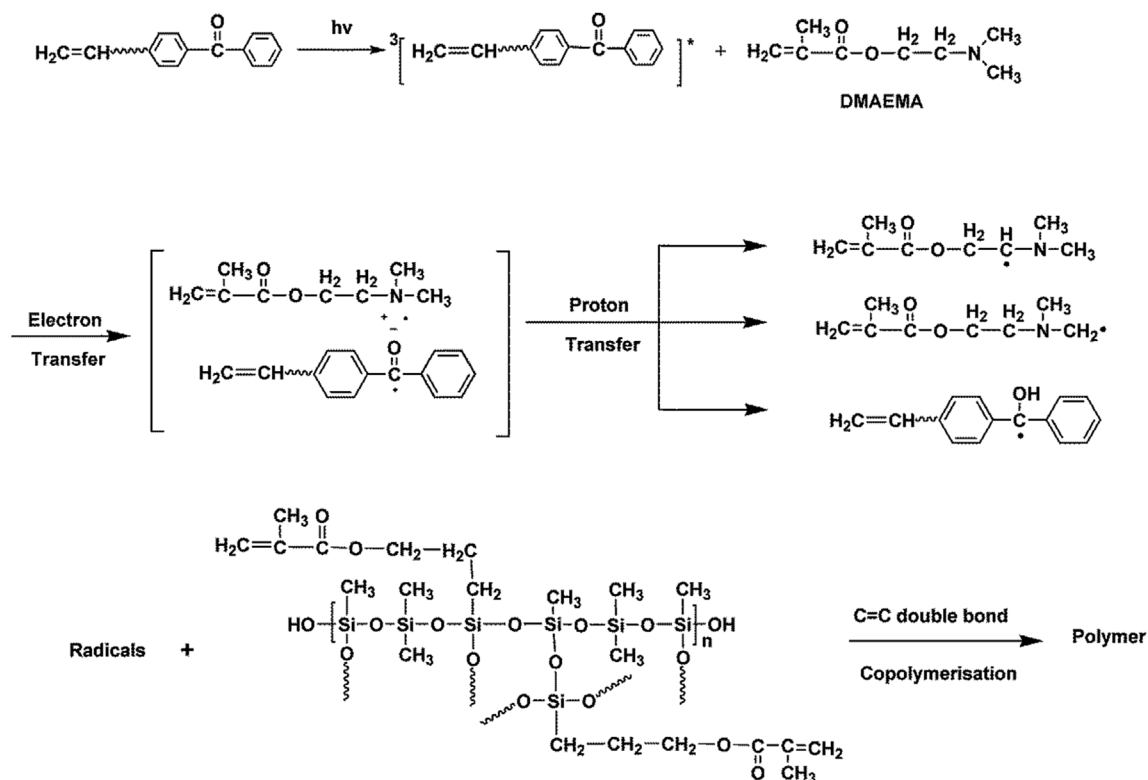


Fig. 6. Photoreaction mechanism of MASR initiated by BPA.

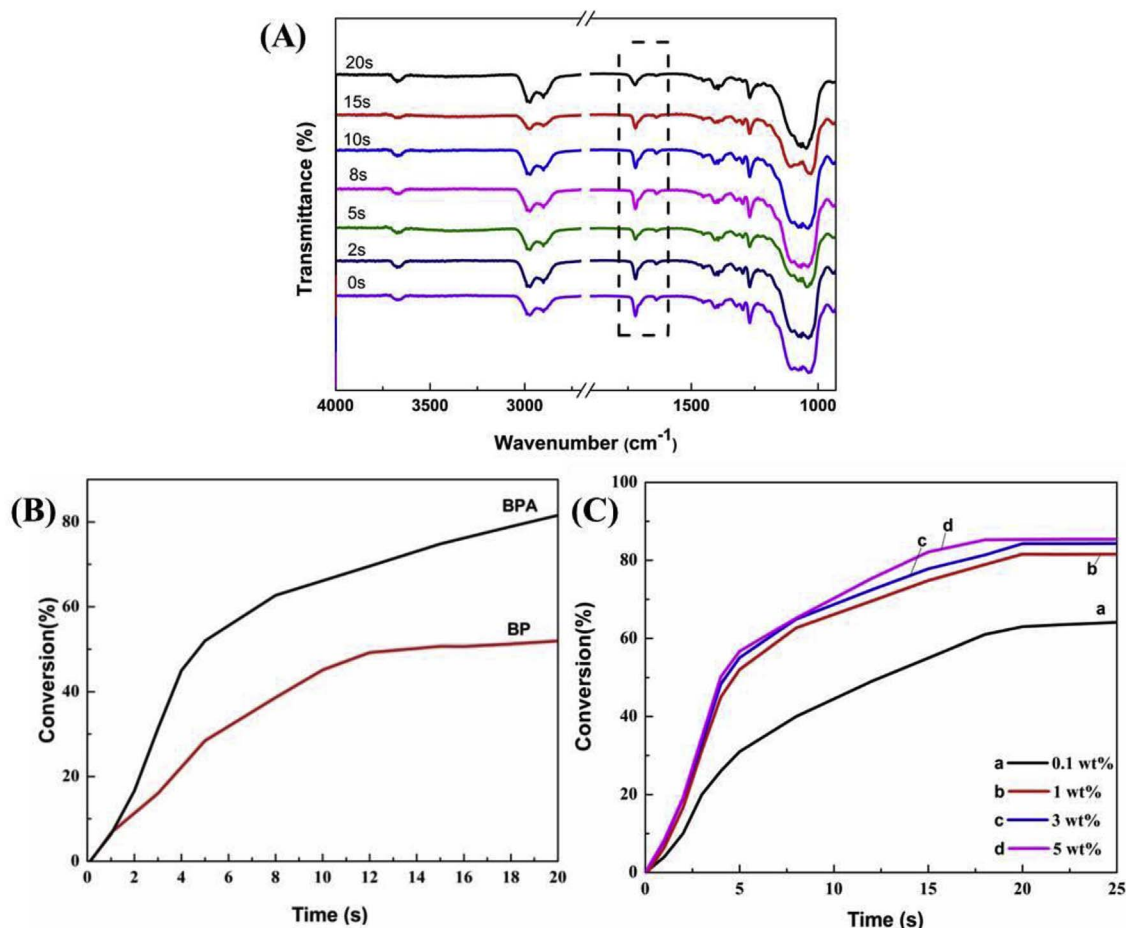


Fig. 7. (A) FTIR spectra of polymerization of MASR initiated by BPA with different UV radiation time. (B) Conversion versus time for photopolymerization of MASR initiated by BPA and BP. (C) Conversion versus time for photopolymerization of MASR using different concentrations of BPA.

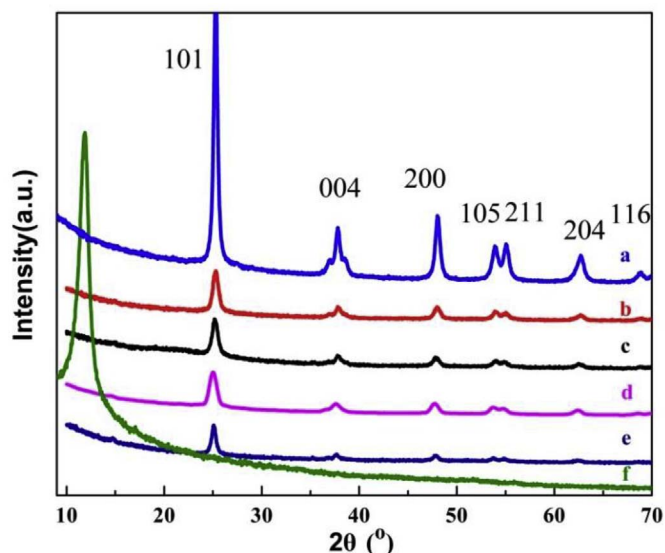


Fig. 8. XRD pattern of a TiO_2 ; b rGO- TiO_2 -1; c rGO- TiO_2 -2; d rGO- TiO_2 -3; e rGO- TiO_2 -4 and f GO.

3.2. UV-curing of MASR based on BPA

Photosensitive methacrylic was incorporated into silicone resin molecular chain. The designed molecular structure of MASR was manifested in Fig. S1. The photolysis of benzophenone could produce a

radical which comes from the carbonyl compound and another radical derives from coinitiators. However, steric hindrance and electronic delocalization make these ketyl-type radicals unreactive agents [25]. The photopolymerization of the MASR containing carbon-carbon double bonds is initiated exclusively by active amine radicals [26], as shown in Fig. 6. BPA containing acryloxy could not only generate radicals from the photolysis reaction, also could be consumed via copolymerizing with MASR during polymerization, which significantly reduces the migration of the active species.

UV curing behaviors of MASR was studied via FTIR spectroscopy as shown in Fig. 7(A). The relative conversion percentage of unsaturated carbon-carbon double bonds in MASR was determined from the integration of the absorption band of the reactive functionality ($\text{C}=\text{C}$ peak at around 1638 cm^{-1}) after normalization by a constant signal in the spectra ($\text{C}=\text{O}$ peak at around 1720 cm^{-1}) [27]. Thus, the conversion of the carbon-carbon double bonds or the extent of reaction could be determined by integrating the peak area according to the following eqn,

$$W = \left\{ 1 - \frac{(A_{1638}/A_{1720})_t}{(A_{1638}/A_{1720})_0} \right\} \times 100\%$$

Where A_0 is the area of absorption peak before UV exposure and A_t is the area of absorption peak after UV exposure.

The double bonds conversion of MASR with different UV exposure time initiated by BP and BPA (1 wt%) was shown in Fig. 7(B). The final conversion degree of BP/DMAEMA photoinitiator system only reaches 52% with 20 s UV radiations. Compared with BP, the BPA is much more efficient, as the final conversion reaches 81% after UV radiation 20 s.

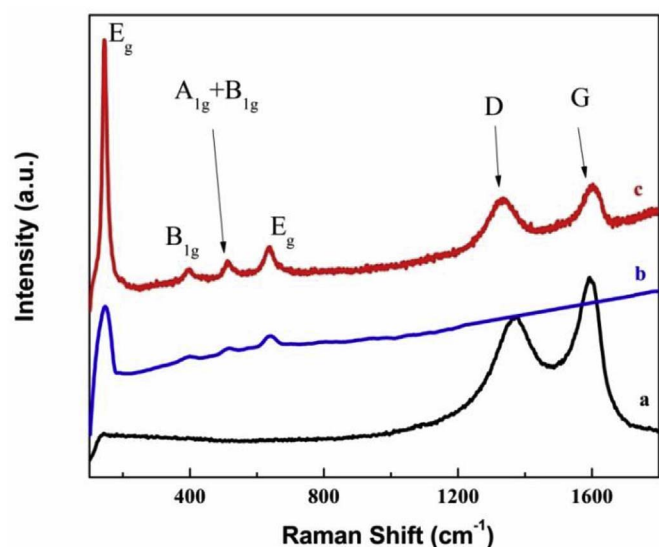


Fig. 9. Raman spectra of a GO; b TiO₂; c rGO-TiO₂.

The high photoefficiency of BPA may be ascribed to the higher electron/proton transfer rate between BPA and DMAEMA and the red-shifted maximal UV absorption of BPA [28]. The polymerizable photoredox system significantly may reduce the migration of the active species, leading to a high efficiency. Fig. 7(C) further illustrates that the

effect of the photoinitiator concentration on the photopolymerization of MASR. With the addition of BPA concentration increases. After 20 s UV exposure, the carbon-carbon double bonds conversion reaches 63% when the concentration of BPA is 0.1 wt%. And the carbon-carbon double bonds conversion increases to 81% with 1 wt% BPA. However, when the concentration of BPA increases from 1 wt% to 5 wt%, the double bonds conversion of silicone resins has little difference. So in this system the optimal concentration of BPA was 1 wt% which was used in the following.

3.3. UV-curing of MASR based on BPA and rGO-TiO₂

RGO-TiO₂ composites were added into MASR to evaluate their influence on UV-curing reactivity and thermal stability. Fig. 8 shows the XRD patterns of the TiO₂, GO and rGO-TiO₂ composites. It can be found that rGO-TiO₂ and pure TiO₂ having a series of diffraction signals at 25.4, 37.7, 48.0, 53.9, 55.1, 62.6 and 68.9° refer to (101), (004), (200), (105), (211), (204), and (116) planes, respectively and is indexed to anatase phase of TiO₂ [29]. Additionally, the intensities of the diffraction peaks present a slight decline with the increasing of GO. The average crystal size of TiO₂ and rGO-TiO₂ are determined through the Scherrer formula to be about 5 nm and 10 nm, respectively [30]. These results prove that TiO₂ doping with GO has little impact on TiO₂ phase structure. Besides, the diffraction peak of GO decreases with the introduction of TiO₂ in the composites, suggesting the conversion of GO to rGO.

As an indicator of existence of TiO₂ and GO in composite material,

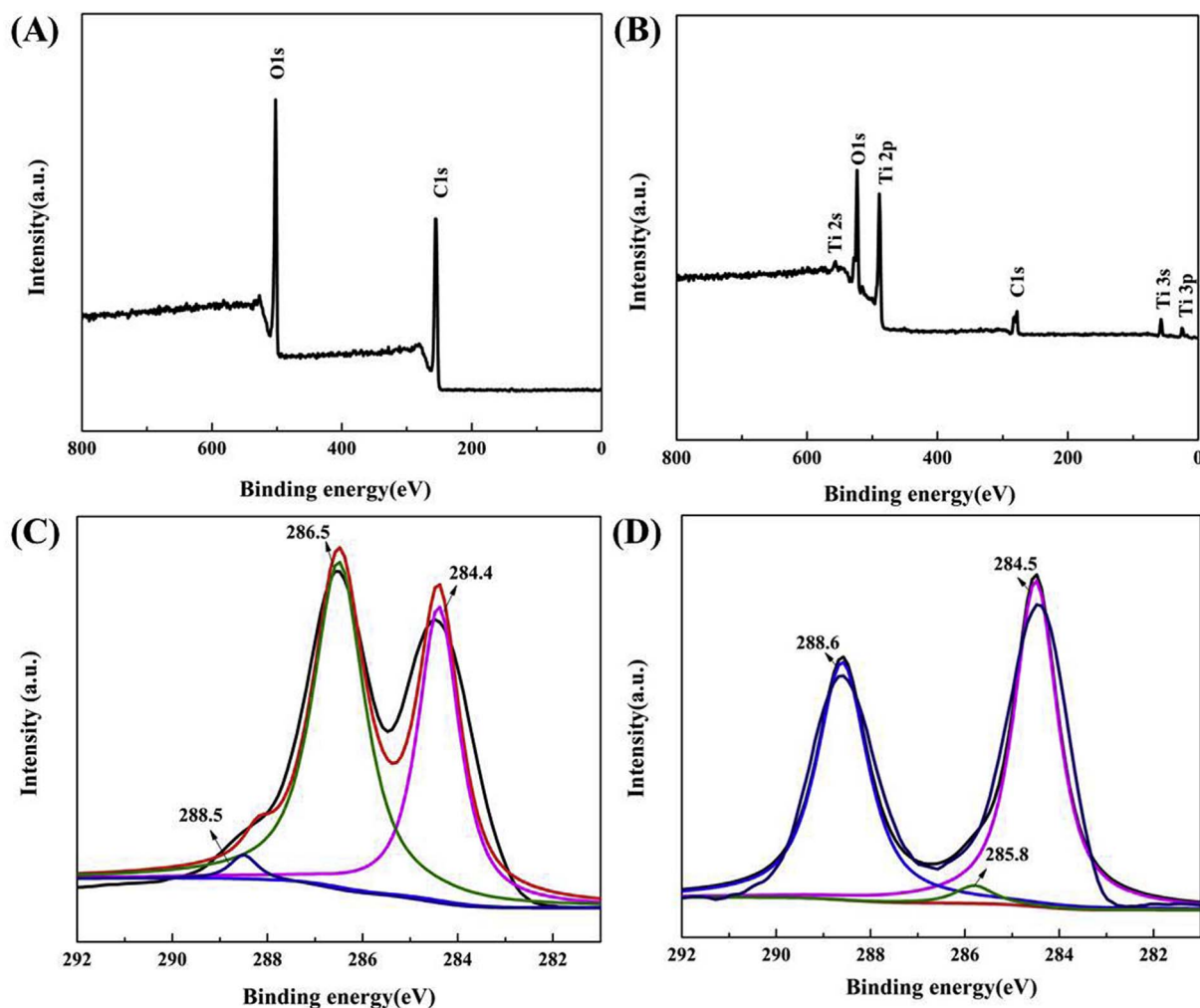


Fig. 10. XPS of (A) wide scan of GO; (B) wide scan of rGO-TiO₂; (C) deconvolution C1s spectra of GO; (D) deconvolution C1s spectra of rGO-TiO₂.

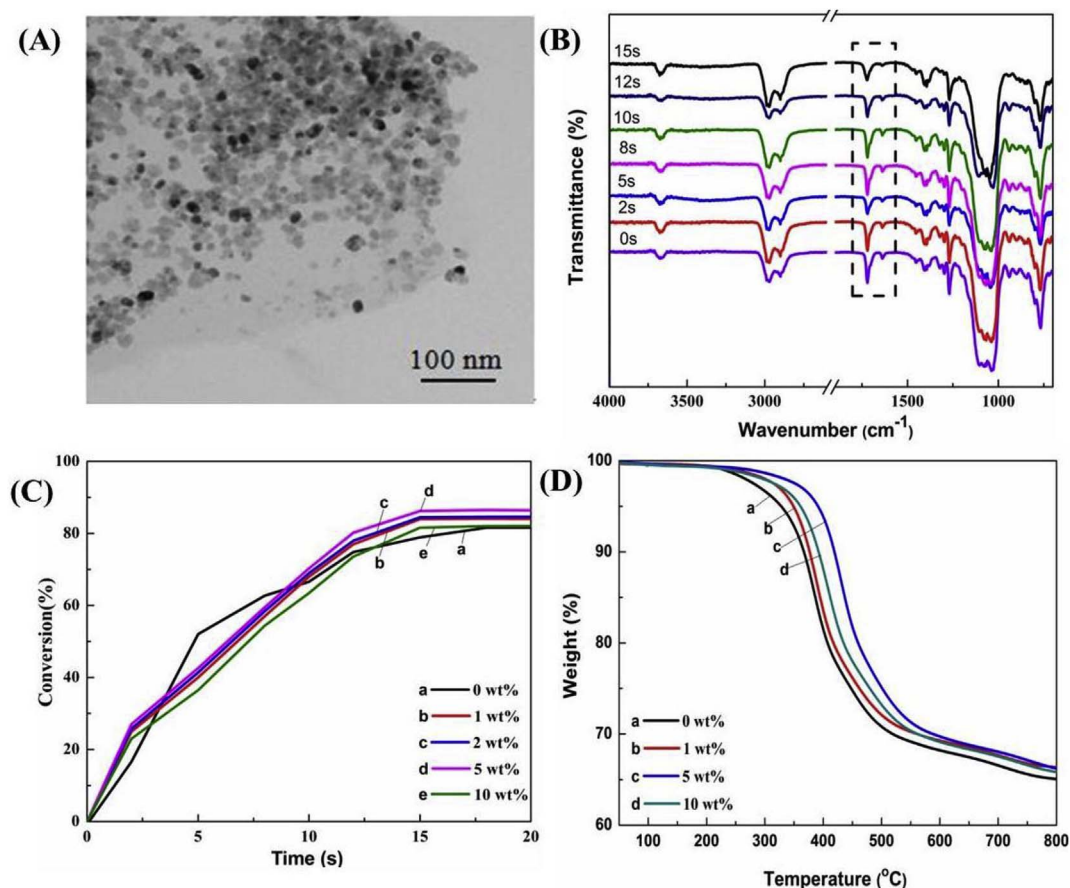


Fig. 11. (A) The microstructure of rGO-TiO₂ (B) FTIR spectra of polymerization of MASR initiated by BPA and rGO-TiO₂ with different UV radiation time. (C) Conversion versus time for photopolymerization of MASR using different concentrations of rGO-TiO₂. (D) TGA curves of MASR with different concentrations of rGO-TiO₂ under nitrogen atmosphere.

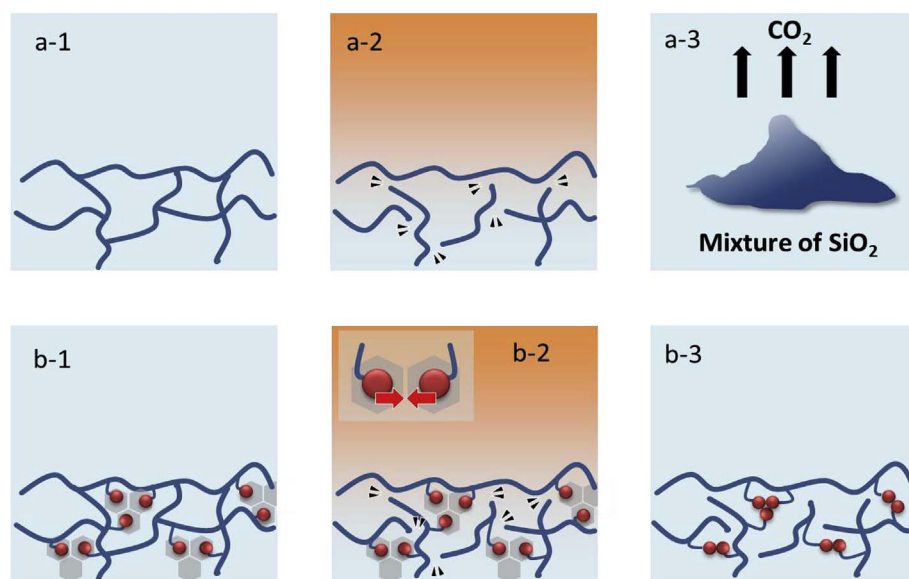


Fig. 12. Degradation sketch of the UV-cured MASR.

the Raman spectra were recorded (see Fig. 9). The four bands located at around 145 (E_g), 396 (B_{1g}), 518 ($A_{1g} + B_{1g}$), and 639 cm^{-1} (E_g) are characteristic for pure anatase TiO₂. The E_g bands are assigned to symmetric stretching vibrations of O–Ti–O, while the A_{1g} and B_{1g} bands refer to the symmetric and asymmetric bending vibration modes of O–Ti–O bonds [31]. Besides, there are two typical characteristic peaks at around 1370 and 1595 cm^{-1} , corresponding to the D and G bands of graphite, respectively [32]. Additionally, the D/G intensity ratio shows

a significant increase for rGO-TiO₂ nanocomposite compared with that of GO, suggesting that the defects are increased after the reduction of GO [33].

To further confirm the structure of rGO-TiO₂, XPS analysis was taken and the spectrum was prepared in Fig. 10. The wide-scan XPS spectrum of rGO-TiO₂ shows that the rGO-TiO₂ contained C, Ti and O. For GO, the main peak observed at 284.4 eV is assigned to the presence of C–C, C=C, and C–H bonds of sp^2 carbon in GO [34]. The binding

energy peaks of 286.5 eV and 288.5 eV are corresponding to C–O and O–C=O, respectively [35]. For rGO-TiO₂, the decrease in C–O peak intensity is an indicator that the oxygen-containing functional groups have been partially reduced.

The microstructure of the rGO-TiO₂ nanocomposite was observed by TEM analysis as shown in Fig. 11(A). The capability of rGO-TiO₂ to promote the curing reaction can be explained by considering that, under UV light, photo-generated electron-hole pairs at the TiO₂ surface. Graphene can play the role of electron receptor and transporter for TiO₂, and the lifetime of electron-hole pair can be enhanced significantly. Therefore, the synergistic effects from graphene and TiO₂ could further enhance the carbon-carbon double bonds conversion in UV-curing silicone resins system initiated by BPA [16].

The behavior of rGO-TiO₂ in MASR and in promoting conversion of its methacrylic moieties was studied by FTIR spectroscopy. The carbon-carbon double bonds conversion in the formulation was followed by monitoring the decrease in the absorbance at 1638 cm⁻¹ which was same with Fig. 7(A). As seen in Fig. 11(C), the presence of more rGO-TiO₂ particles involves an increase in the carbon-carbon double bonds conversion of MASR. In fact, the maximum conversion value has been recorded 86% at a 5 wt% rGO-TiO₂ loading after 15 s UV radiations, with an increase of 6.2% compared to BPA triggered system (20 s). However, more rGO-TiO₂ (10 wt%) results in a decreased conversion. The conversion curves confirm that a higher photopolymerization conversion is achieved in shorter reaction time for the formulation containing the rGO-TiO₂ with respect to the above sample (Fig. 7(C)), demonstrating that the embedded rGO-TiO₂ significantly increase UV-curing reactivity of the neat organic formulation.

Thermal stability of the UV-cured MASR and its subsequent nanocomposites were evaluated through the determination of the degradation temperature at 5% weight loss ($T_{5\%}$) and the residual weight at 800 °C (Y_c). Fig. 11(D) shows the TGA curves of all the samples. The $T_{5\%}$ and Y_c of the unfilled resin are 327.1 °C and 65.1%, respectively. These parameters are all increased with the increase of the rGO-TiO₂ amount, reaching their highest values with 5% rGO-TiO₂. In fact, $T_{5\%}$ increases to 389.6 °C and Y_c at 800 °C reaches a value of 66.2%. The enhancements seen in $T_{5\%}$ are related to the shielding effect of the rGO-TiO₂ that act as a barrier reducing the motion and insulating the materials from the heat source [36,37]. However, the $T_{5\%}$ and Y_c of the MASR decrease when the rGO-TiO₂ increases to 10%, which may be arising from the aggregation of nanoparticles and consequently leads to poorly disperse in the polymer matrix.

The degradation sketch of the MASR was shown in Fig. 12. Fig. 12(a-1) shows the molecular chair of the MASR. At high temperature, the Si–OH end groups have reacted and the Si–CH₃ groups in the resin are oxidized, releasing CO₂ and H₂O while SiO₂ remained in the crucible (Fig. 12(a-2) and Fig. 12(a-3)) [38]. However, the existence of rGO-TiO₂ significantly affects the degradation of the polymer matrix. When temperature increased, nanostructure tends to incorporate into the silicone molecular chain and avoid the condensation of the Si–OH. Thus, rGO-TiO₂ nanostructure on the silicone molecular chain plays a critical role in preventing the degradation of polymer molecular chain.

4. Conclusion

In summary, we have reported an efficient technique to enhance the polymerization degree of photosensitive methacrylic-silicone resins via polymerizable photoinitiator and rGO-TiO₂. The results indicate that BPA containing acryloxy is an efficient photoinitiator, which could generate radicals from the photolysis reaction and be consumed via copolymerizing with the methacrylic-silicone resins. The UV-cured MASR possesses a high carbon-carbon double bonds conversion above 80% initiated by BPA after 20 s UV radiations. Furthermore, the synergistic effects from graphene and TiO₂ further enhance the carbon-carbon double bonds conversion to 86% in shorter reaction time (15 s). Besides, the rGO-TiO₂ improved the thermal stability of the UV-cured

MASR. Based on these findings, it can be concluded that the designed photoinitiator and rGO-TiO₂ nanocomposites could be applied in the photopolymerizable silicone resin and UV-curing fields.

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Appendix A. Supplementary data

Supplementary data related to this article can be found at <http://dx.doi.org/10.1016/j.compositesb.2017.12.006>.

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