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1 **Thermal behaviour and fungi resistance of composites based on wood and natural**
2 **and synthetic epoxy resins cured with maleopimaric acid**

3
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8
9 **Abstract**

10 Wood/epoxy resin composites based on diglycidyl ether of bisphenol A (DGEBA)/epoxidized oil
11 (EO)/Diels–Alder adducts of resin acids with maleic anhydride (RAMA) in acetone (80 %) were obtained by
12 impregnation. Epoxidation was carried out in the presence of hydrogen peroxide (H₂O₂), acetic acid (AAc),
13 sulfuric acid (H₂SO₄) as catalyst and cyclohexane (CHx) as solvent at a molar ratio of 0.5/1.5/1
14 (AAc/hydrogen peroxide/ethylene unsaturation). The kinetic parameters (activation energy and pre–
15 exponential factor) of the DGEBA/EO/RAMA curing reaction and thermal characterization of the
16 crosslinked compounds and wood epoxy composites (WECs) were obtained using differential scanning
17 calorimetry (DSC) and thermogravimetric analysis (TGA). Based on the chemical structure of the evolved
18 gases resulted during thermal decomposition of the crosslinked polymers and WECs, identified with FT–IR,
19 a probable mechanism of thermal degradation was proposed. Resistance to three fungi was also tested.

20
21 **Keywords:** Wood, Thermosetting resin, Cure behaviour, Thermal properties, Biological resistance

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1 **1. Introduction**

2 Nowadays, composites based on natural materials are widely used in industrial and household fields,
3 because they are environmentally friendly and originate from renewable resources. Wood is among the most
4 important renewable resources. It has a great importance for life sustainability, as well as both raw and
5 processed material. The most important components of wood are lignin, cellulose, hemicelluloses and resins.
6 Under environmental factors influence (i.e. temperature, light, moisture, microorganisms, insects, etc.), there
7 occurs wood damage, followed by a decrease in technical (thermal, mechanical and electrical behaviour) and
8 aesthetic aspects [1, 2].

9 To preserve the properties of wood, physical and chemical methods, including heat treatment,
10 chemical modification, impregnation and physical coatings, have been used. Considering its chemical
11 structure (mainly hydroxyl groups), wood was modified by esterification and etherification reactions with
12 anhydrides, acids, acid chlorides, epoxy resins, unsaturated derivatives, cyclic ethers, lactones, alkyl
13 chlorides and so on. These transformations were aimed to improve mechanical and electrical properties, as
14 well as to enhance resistance to environmental factors action [3–13].

15 Vegetable oils (edible or non-edible) are a renewable source of raw materials, mainly esters of
16 glycerol, which are the result of glycerol esterification with three saturated and/or unsaturated fatty acids [14].
17 As a consequence of the chemical structures, these oils may be modified to obtain new monomers that can
18 induce special properties in the final polymers (flexibility, biocompatibility, biodegradability, thermal
19 stability) [4, 15–20]. Hemp oil and flax oil, as non-edible oils that contain a large number of unsaturated
20 groups in their structure, represent an excellent source of raw material for the synthesis of monomers with
21 new functional groups capable to polymerize. One of the most common methods of chemical modification of
22 vegetable oils is epoxidation. Epoxidized oils may be used as basis for their transformation in acrylated,
23 methacrylated, hydroxylated monomers, used either individually or mixed in various polymeric compositions
24 [16, 17, 21–25]. Also, vegetable oils and their derivatives, alone or in combination with various fungicides,
25 have been used to increase the decay resistance of wood [26, 27].

26 Rosin is a renewable naturally raw material consisting of neutral compounds (turpentine) and resin
27 acids (abietic, levopimaric, neoabietic, palustric, fatty acids, etc.). As a consequence of their chemical
28 structure, the resin acids can be converted into several types of monomers. In particular, the most common
29 transformation is the Diels–Alder reaction, when some polyacids are obtained and further used as

1 crosslinking agents for epoxy resins [28–35]. These adducts, through their chemical structure, may improve
2 the properties of the crosslinked epoxy resins.

3 Epoxy resins are some of the most widely available synthetic monomers, which, after curing,
4 present excellent electrical, mechanical and thermal properties in a vast palette of application fields, although
5 not being environmentally friendly [36, 37]. This behaviour depends on the formulation, chemical
6 composition and crosslinking conditions. By using some natural ingredients (rosin acids and vegetable oil
7 derivatives), these formulations may compensate some drawbacks and be environmentally friendly [16, 20,
8 28–35, 38, 39].

9 The paper describes the curing and non-isothermal degradation kinetics of compositions based on
10 DGEBA/EO/RAMA and WECs. The novelty of the work consists in the using of some epoxy resins based
11 on vegetable oils and of a curing agent derived from resinic acids of natural origin. Also, the thermal and
12 fungal resistance properties were followed using DSC and TG analyses. Based on FT-IR analysis of the
13 evolved gases during thermal decomposition a possible degradation mechanism is proposed.

14

15 **2. Experimental**

16 *2.1. Materials*

17 Hydrogen peroxide (H_2O_2), glacial acetic acid (AAc), maleic anhydride (MA), potassium hydroxide
18 (KOH), sulfuric acid (H_2SO_4), triethylbenzylammonium chloride (TEBAC) and anhydrous sodium sulfate
19 (Na_2SO_4) were analytical grade products and used without any purification. The solvents were used as
20 received or distilled before use. Resin acids (RA) were obtained from commercial rosin by acid
21 isomerization and by repeated recrystallizations from concentrated acetone solutions and had an acid number
22 (a. n.) of $181 \text{ mg KOH g}^{-1}$ and melting point (m. p.) of $161 \text{ }^\circ\text{C}$ [33]. RAMA with m. p. $230 \text{ }^\circ\text{C}$ and a. n. 434
23 mg KOH g^{-1} was obtained following a procedure described in the literature [40]. DGEBA, a commercial
24 product (Sintofarm, Romania), (average epoxy equivalent weight of 190 g eq^{-1} , viscosity of 12000 mPas at
25 $25 \text{ }^\circ\text{C}$) was used as received. Flax oil (FO) and hemp oil (HO) have been obtained from the local market and
26 have the following compositions supplied by the manufacturers: flax oil [saturated fats $10.1 \text{ wt. } \%$,
27 unsaturated fats (oleic acid $22.4 \text{ wt. } \%$, linoleic acid $14.3 \text{ wt. } \%$, α -linolenic acid $52.3 \text{ wt. } \%$), other acids 0.9
28 $\text{wt. } \%$] and hemp oil [saturated fats $8 \text{ wt. } \%$, unsaturated fats (oleic acid $11.4 \text{ wt. } \%$, linoleic acid $57 \text{ wt. } \%$,
29 α -linolenic acid $18.7 \text{ wt. } \%$, other acids $2.9 \text{ wt. } \%$)]. Epoxidized flax oil (EFO) (number of epoxy groups 5.9
30 mol^{-1}) was obtained as described below and epoxidized hemp oil (EHO) (number of epoxy group 3.5 mol^{-1})

1 was obtained following a procedure described in the literature [23]. The number of epoxy groups was
2 measured using $^1\text{H-NMR}$ spectroscopy. The softwood samples (W) (*Abies alba L.*) had the same
3 characteristics as those presented in a previous work [9].

4

5 2.2. Sample preparation for DSC and TGA studies

6 For the DSC studies, the formulation DGEBA/EO/RAMA at a molar ratio of 0.8/0.2/0.66 (epoxy
7 ring/carboxyl proton 1/1) in the presence of TEBAC as catalyst was prepared. After a vigorous mixing of the
8 reactants, they were degassed in a vacuum oven at room temperature and a portion has been stored at 0 °C
9 before the DSC recording was performed. The DGEBA/EO/RAMA compositions used for the TG
10 characterization were transferred in an aluminum form and crosslinked according to the following protocol: 1
11 h at 100 °C, 1 h at 135 °C and post cured 4 h at 185 °C (Scheme 1).

12

Scheme 1

13 The WEC samples were obtained by immersion of W samples for 8 h in the next formulation:
14 DGEBA/EO/RAMA at molar ratio 0.8/0.2/0.66, 80 wt. % solution in acetone, in the presence of TEBAC as
15 catalyst. After immersion, the impregnated samples were placed in a vacuum oven at 40 °C and maintained
16 another 8 h to release the solvent. Then, the WECs were cured at 70 °C for 1 h, 1 h at 100 °C, 1 h at 135 °C
17 and then post cured at 150 °C for 3 h and used for TGA studies. The same samples were also used for
18 crystallinity measurements and for morphological analysis. The weight fraction of W in WECs was 90 %.

19

20 2.3. Measurements and methods

21 The acid number (a.n.) was obtained using a slightly modified ASTM D 465–05. The sample
22 dissolved in acetone at 10 % (w/v) concentration was titrated with 0.5 N ethanolic KOH solutions in the
23 presence of phenolphthalein as indicator [35, 41]. The epoxy equivalent weights (EEW) were obtained using
24 ASTM D 1652–04 [42]. The number of the double bonds from oils was obtained using $^1\text{H-NMR}$
25 spectroscopy, with the next equation [43]:

26

$$D_b = 0.5(A - N_f) / N_f \quad (1)$$

27

28

29

where: D_b is the double bonds number, A is the surface
assigned to the olefinic protons located between 5.2 and 5.4 ppm and N_f is the normalization factor (obtained
through the dividing by 4 of the protons area from 4.0 and 4.3 ppm assigned to terminal protons of glycerol).

1 The melting points were obtained with a DSC 200 F3 Maia device (Netzsch, Germany), at 10 °C min⁻¹
 2 heating rate.

3 FT-IR spectra were acquired with a Vertex 70 apparatus (Bruker–Germany) using KBr pellets. ¹H–NMR
 4 spectra of DGEBA, oils, epoxidized oils and RAMA were recorded with an Avance DRX 400 (Bruker–
 5 Germany) apparatus at 50 °C using CDCl₃ and DMSO–d₆ as solvents and tetramethylsilane (TMS) as
 6 internal standard.

7 TG curves were recorded in the temperature range 30–600 °C, in inert atmosphere (N₂ flow rate of 40 mL
 8 min⁻¹) at three heating rates (10, 20 and 30 °C min⁻¹) using STA 449 F1 Jupiter device (Netzsch–Germany).
 9 Open Al₂O₃ crucibles were loaded with approximately 10 mg of the ground sample. Gaseous products
 10 resulting from thermal degradation were transferred through a transferring line made from Teflon, coupled to
 11 the head of the TG balance, with 1 m length and with 1.5 mm diameter (heated at 190 °C to limit
 12 condensation) to the FTIR apparatus having a MCT detector (Mercury Cadmium Telluride). The FT–IR
 13 spectra were recorded between 4000 and 500 cm⁻¹ wavenumber values and processed with OPUS 6.5
 14 software.

15 DSC analysis was undertaken on a DSC 200 F3 Maia apparatus (Netzsch, Germany), under nitrogen flow
 16 rate of 50 mL min⁻¹, at three heating rates (5, 10 and 15 °C min⁻¹) in the temperature range 25–300 °C. The
 17 calibration of the apparatus was made with high purity indium. The samples weighing up to 10 mg were
 18 placed in aluminum cells provided with a pierced lid, were scanned in the presence of one aluminum empty
 19 cell as standard. The heat flow versus temperature was recorded. DSC data were recorded consecutively
 20 using the same sample on the same day [44].

21 The evaluation of the kinetic parameters, both for curing and thermal degradation reactions were obtained
 22 using DSC and TG scans and processed with Netzsch Thermokinetics 3 software [45]. In a first instance,
 23 these parameters were evaluated on the basis of free–estimation methods of Ozawa–Flynn–Wall (OFW) and
 24 Friedman (FD). The used equations are as follows [46–48].

$$25 \quad \ln\beta = C - 1.052(E_0/RT_p) \quad (\text{OFW equation}) \quad (2)$$

26 and

$$27 \quad \ln\left(\frac{d\alpha}{dT}\right) = \ln\left(\frac{A}{\beta}\right) + \ln(f(\alpha)) - \frac{E_{FD}}{RT} \quad (\text{FD equation}) \quad (3)$$

1 where: A is the frequency factor, β is the heating rate, C is a constant, E_O and E_{FD} are the specific activation
2 energies of reactions obtained with OFW and FD relations, T_p is the exothermic peak temperature in Kelvin
3 and R is universal gas constant.

4 These methods, included in “Netzsch Thermokinetics 3”, a software module for the kinetic analysis of
5 thermal measurements, permit the evaluation of the kinetic parameters and the variation of activation energy
6 values versus conversion degree, without assuming any reaction model. Changes in the shape of the
7 activation energy versus the degree of conversion curves indicate that the degradation reactions occur
8 following a complex mechanism in several independent stages through consecutive, parallel or competitive
9 processes [49, 50]. This makes it possible to choose between different mechanisms and obtain a global model
10 that gives the best results for the curing or thermal degradation processes.

11 Morphological study of WECs was conducted by analyzing the scanning electron microscopy (SEM) images.
12 These were obtained using the SEM/ESEM–EDAX–QUANTA 200 apparatus which operated in the
13 following parameters: field emission filament 16 kV accelerating voltage, high vacuum mode and maximum
14 magnification of 5000 X. The images were obtained on the surface samples, cut perpendicularly on the sense
15 of growth and gilded to make them conductive.

16 Anti–decay resistance testing was performed on untreated wood (blank), polymer and WECs. The test
17 samples were obtained as previously described above. The main components of wood (cellulose,
18 hemicelluloses, lignin), under favorable conditions of humidity and temperature, can be metabolized by fungi
19 as nutrients, resulting in wood degradation. Four samples from each specimen, dried to constant weight at
20 105 °C, were tested in the same conditions for resistance against decay. The following fungi were used: a
21 white rot fungus *Penicillium chrysogenum* ATCC10106 (*P. chrysogenum*), a brown rot fungus *Cladosporium*
22 *cladosporioides* ATCC16022 (*C. cladosporioides*) and black rot fungus *Aspergillus brasiliensis* ATCC 9642
23 (*A. brasiliensis*). The samples were placed in Petri dishes on a substrate of malt extract (for *P. chrysogenum*)
24 and potato dextrose (for *C. cladosporioides* and *A. brasiliensis*) in agar which were previously inoculated
25 with a standard inoculum of 0.5 McFerland for each fungal strain. All the Petri dishes were incubated at the
26 room temperature (24 ± 1 °C) for 10 weeks. The fungal growth on the untreated and treated samples was
27 monitored during the 10 weeks of incubation and at the end of the experiment the wood samples were
28 removed from the culture plates and the fungal biomass was gently removed with the aid of a wet cotton
29 fabric. The wet samples were oven–dried at 105 °C, in order to reach constant weight. Then all the samples
30 were accurately weighed in order to notice the retained moisture.

1 Crystallinity of W and WECs was measured with the aid of wide angle X-ray diffractometry (WAXD). The
 2 WAXD curves were recorded using a Bruker D8 Advance (Germany) apparatus with $\text{CuK}\alpha$ monochromatic
 3 radiation ($\lambda = 0.15406$ nm) at 25 °C, 36 kV and 25 mA, in the interval 2θ diffraction angles (Bragg angle)
 4 ranging between 4 and 40 °.

5 The crystallinity index was calculated using the next equation [51]:

$$6 \quad I_c (\%) = [(I_{002} - I_{am})/I_{002}] \cdot 100 \quad (4)$$

7 where: I_c is crystallinity index, I_{002} is maximum value of intensity at the 002 crystalline peak, I_{am} is the
 8 maximum value of amorphous peak intensity [52].

9 The polymer weight percentage gain (WPG) after crosslinking was calculated with the next equation:

$$10 \quad \text{WPG} (\%) = [(M_f - M_i)/M_i] \cdot 100 \quad (5)$$

11 where: M_i is the initial weight of W and M_f is the final weight of WEC sample after crosslinking.

12 The humidity values after fungal attack were calculated as follows:

$$13 \quad H (\%) = [(M_w - M_d)/M_w] \cdot 100 \quad (6)$$

14 where: H is the percent of the humidity; M_w is the weight of wet sample after fungal attack and M_d is the
 15 weight of the same sample after drying.

16 The weight losses (WL) (%), after the fungus attack were calculated with the equation:

$$17 \quad \text{WL} (\%) = [(M_i - M_d)/M_i] \cdot 100 \quad (7)$$

18 where : M_i is the weight of initial sample before fungal attack and M_d is the weight of the same dried sample
 19 after fungal attack.

20 Twenty two solvents with solubility parameters (δ) between 7.24 up to 23.5 cal cm^{-3} were used to obtain the
 21 swelling parameter of the crosslinked polymers (Table 1). Each solvent used in the study was introduced in
 22 Erlenmeyer flasks equipped with plunger stoppers. Over each solvent, polymer samples with a weight of up
 23 to 0.1 g were added, maintaining the constant temperature (25 °C) for a long period of time. The weight of
 24 the swollen samples was measured until it was stabilized. The swelling degree was calculated with the
 25 following equation [53]:

$$26 \quad Q (\%) = [(M_w - M_i)/M_i] \cdot 100 \quad (8)$$

27 where: Q is the swelling coefficient, M_w is the weight of the swollen sample and M_i is the weight of the initial
 28 sample.

29 **Table 1**

30 **Fig. 1_revised**

1 The solubility parameter of the crosslinked samples is obtained by observing the maximum of the swelling
 2 coefficient (Q) in the Q versus δ plot (Fig.1). As it may be seen from Fig. 1, the maximum value is reached
 3 for methylene chloride, which is the solvent that induces the highest degree of network swelling. It can be
 4 considered that the solubility parameter of methylene chloride, of $9.93 \text{ (cal cm}^{-3}\text{)}^{0.5}$, may be attributed to the
 5 polymer–solvent interaction parameter of the crosslinked polymers. The molecular weight between two
 6 crosslinking points (Mc) and the crosslinking density (n) of DGEBA/EO/RAMA samples were obtained
 7 based on the swelling measurements and calculated with Flory–Rehner equation [54]:

$$8 \quad n = -[\ln(1-V) + V + \chi \cdot V^2] / V_{mol} [V^{0.33} - V / 2] \quad (9)$$

9 where: V is the volume fraction of polymer in swollen polymer, V_{mol} is the molar volume of the solvent and
 10 χ is the Flory solvent–polymer interaction parameter. The interaction parameter may be obtained with the
 11 equation:

$$12 \quad \chi = \frac{V}{RT} (\delta_p - \delta_s)^2 \quad (10)$$

13 where: R is the gas constant, T is temperature in Kelvin, V is the molar volume of the solvent, δ_p is
 14 solubility parameter of polymer and δ_s is solubility parameter of solvent.

16 3. Results and discussion

17 3.1. Synthesis and characterization of epoxidized flax oil (EFO)

18 Into a 0.5 L four-necked reaction vessel placed in a water bath, equipped with mechanical stirrer,
 19 thermometer, a water reflux condenser and dropping funnel, 150 g FO, 120 g AAc, 2 mL H_2SO_4 (diluted in 4
 20 mL of distilled water) and 40 mL CHx were introduced. CHx was used to reduce the viscosity of the mixture
 21 and to minimize the epoxy cycle opening. The reaction mixture was moderately stirred for about 30 minutes,
 22 purged with a slow flow of nitrogen and the water bath was cooled with ice to 10 °C. After 1 h, 220 g H_2O_2
 23 30 % (w/w) was added in small portions under vigorous stirring, while maintaining the reaction mixture
 24 temperature at about 15 °C. After adding the entire amount of H_2O_2 , the temperature of the reaction mixture
 25 was slowly raised to 55 °C and maintained at this level for another 6 h. For finishing the reaction, the
 26 mixture was heated to 65 °C for another 0.5 h and then cooled rapidly on ice water bath. The reaction mass
 27 was washed twice with distilled water at 35 °C and then with a 5 wt. % potassium carbonate solution until
 28 complete neutralization. Then, the water layer was discharged and EFO was kept on anhydrous sodium

1 sulfate to remove traces of water. Sodium sulfate was removed by filtration and CH_x was removed at 60 °C
2 in a rotary evaporator under vacuum. The final product was introduced in a polyethylene vessel and stored in
3 the dark at 20 °C (Scheme 2).

4 **Scheme 2**

5 The obtained EFO was a low viscosity yellowish color liquid. Its chemical structure was confirmed
6 using FT-IR and ¹H-NMR spectroscopy analysis. Since there are no significant differences between the FT-
7 IR spectra of the two oils, only the spectra of flax oil and epoxidized flax oil are shown in Fig. 2. The main
8 signals are: 3470 cm⁻¹ (OH stretching from opened epoxy ring), 2921 and 2859 cm⁻¹ (CH₃, CH₂, CH
9 stretching vibrations from fatty acids moieties), 1742 cm⁻¹ (CO stretching in ester groups), 1455 cm⁻¹ (CH
10 aliphatic), 1242 and 1157 cm⁻¹ (C-O-C ester linkages) and 823 cm⁻¹ (oxirane rings). The signals from 3010
11 cm⁻¹ and 1654 cm⁻¹ from FO (specific to the olefin double bond) disappear and a new signal appears at 823
12 cm⁻¹ specific to the epoxide groups from epoxidized oils (Fig. 2b). ¹H-NMR spectra of FO and EFO are
13 presented in Fig. 3. The main signals are: 5.182 (the proton located at the secondary C of glycerol), 4.130 -
14 4.290 (glycerol protons), 2.805 - 3.091 (epoxy group protons), 2.28 (methylene protons adjacent to ester
15 groups), 1.25 - 1.47 (methylene protons); 0.85 - 0.92 (methyl protons). The main difference which appears
16 between the ¹H-NMR spectra of FO and EFO is the absence of the olefinic protons located near 5.35 ppm
17 and presence of the protons attached to the epoxy ring with the peaks located at 2.770 and 3.091 ppm
18 chemical shift.

19 **Fig. 2_revised**

20 **Fig. 3_revised**

21 *3.2. FT-IR characterization of crosslinked polymers and WECs*

22 In Fig. 4 there are presented the FT-IR spectra of W, crosslinked DGEBA/EFO/RAMA and WECs.
23 As it may be observed in the DGEBA/EFO/RAMA and WECs spectra, they do not show any signal specific
24 to the epoxy rings (at 915 cm⁻¹ for DGEBA and at 823 cm⁻¹ for EFO), confirming the reaction between the
25 epoxy and carboxyl groups from RAMA. As a result of this reaction, a number of new hydroxyl groups are
26 generated. Their corresponding signals are located at 3346 cm⁻¹ in the WEC spectrum (Fig. 4c) and overlap
27 with the signal from 3343 cm⁻¹ specific to wood (Fig. 4a). Also, due to the above reaction, new ester groups
28 appear, which are highlighted by the signal at 1724 cm⁻¹ (C=O stretching vibrations) and signals from 1234
29 cm⁻¹ and 1028 cm⁻¹ (C-O-C stretching vibrations) (Figs 4b and 4c). There may also be observed a series of
30

1 signals located near 1606 and 1509 cm^{-1} , specific to skeletal vibrations of aromatic entities from DGEBA
2 (Figs 4b and 4c).

3 **Fig. 4_revised**

4

5 *3.3. Crosslinking density measurements*

6 The crosslinking density (n) and the molecular weight between two crosslinking points (M_c) of
7 DGEBA/EO/RAMA networks were determined using swelling measurements and a method described in the
8 literature [54]. Based on the data from Fig. 1, the solubility parameter of methylene chloride [9.93 (cal cm^{-3})
9 $^{0.5}$] can be assigned as the polymer–solvent interaction parameter of the two crosslinked polymers. The
10 following values were obtained: $n = 2.065 \cdot 10^{-3} \text{ mol cm}^{-3}$ and $M_c = 591 \text{ g mol}^{-1}$ for DGEBA/EHO/RAMA
11 sample and $n = 1.862 \cdot 10^{-3} \text{ mol cm}^{-3}$ and $M_c = 633 \text{ g mol}^{-1}$ for DGEBA/EFO/RAMA. These values of M_c are
12 quite close to the theoretical ones, suggesting a random distribution of the epoxidized oil molecules in the
13 crosslinked polymer structure. The lowest value of M_c for DGEBA/EHO/RAMA sample is due to the
14 smaller number of epoxy groups from EHO, fact which leads to a smaller degree of network packaging. The
15 obtained networks have relative low T_g values (31 °C for DGEBA/EHO/RAMA and 32 °C for
16 DGEBA/EFO/RAMA), as a consequence of internal plasticizing effect introduced by epoxy oils.

17

18 *3.4. Curing of DGEBA/EO with RAMA*

19 The obtained DGEBA/EO/RAMA samples (epoxy ring/carboxyl proton 1/1) were crosslinked in the
20 DSC device at three heating rates. In Scheme 1 the chemical reactions between carboxyl and the epoxy
21 groups from DGEBA/EO/RAMA are presented. Since the main reagents in these formulations have more
22 than two reactive groups, with time and temperature increase, three–dimensional networks were obtained. As
23 one may see from Fig. 5a (DSC curves for DGEBA/EFO/RAMA), with the increase in temperature, besides
24 the main peak, a smaller sized peak appears, suggesting that few of the EFO epoxy groups (which have 6
25 epoxy groups) are partially sterically hindered, these groups further reacting with temperature increase. In the
26 case of DGEBA/EHO/RAMA (Fig. 5b), the DSC curves show the same trend at 5 °C min^{-1} , while the records
27 at high speeds, present only an exothermic peak, suggesting that all chemical reactions occur simultaneously.
28 The evaluation of the kinetic parameters of the crosslinking reactions (apparent activation energy and
29 preexponential factor) of the above samples was undertaken using the Friedman (FD) and Ozawa (OZ)
30 methods [45–48].

Fig. 5_revised

1
2 If the curves shapes of apparent activation energy as a function of conversion factor show deviations from
3 linearity, this fact suggests that the crosslinking takes place in several steps (maybe consist of a series of
4 successive or parallel processes) and occur following a complex mechanism [49, 50]. From Figs. 6a and b, it
5 may be observed that the apparent activation energy curves present few maximum and minimum values and
6 depend on the chemical composition of the samples. This suggests that in order to establish the mechanism
7 of the crosslinking process it is necessary for the evaluation to be done using the non-linear regression
8 method (MLR) within the “Thermokinetics 3” software. Based on the initial values of the kinetic parameters
9 obtained with the FD method and the shape of the DSC curves recorded at three heating rates, the software
10 solves numerically the differential equations specific to 18 reaction types, adjusting the theoretical models to
11 registered data. Following the processing of data by the software in the 0.1 to 0.9 conversion degree (α)
12 interval, a three step kinetic model with successive reactions for the DGEBA/EFO/RAMA sample and a two
13 steps model with successive reactions for the DGEBA/EHO/RAMA sample were chosen. The choice of
14 these models was made using the difference between measured and calculated values, on the basis of the F-
15 test and the best correlation coefficient between data.

16 The next conversion functions were used:

17 – n^{th} reaction order of the autocatalytic model, Cn:

$$18 \quad f(\alpha) = (1-\alpha)^n (1+K_{\text{cat}} \cdot \alpha) \quad (11)$$

19 where: K_{cat} is the autocatalytic constant, n is the reaction order

20 – reaction order model, Fn:

$$21 \quad f(\alpha) = (1 - \alpha)^n \quad (12)$$

22 where n is the reaction order and α is the conversion degree,

23 and

24 – Avrami–Erofeev reaction model, An:

$$25 \quad f(\alpha) = n(1-\alpha)[- \ln(1-\alpha)]^{\frac{n-1}{n}} \quad (13)$$

26 where n is a constant parameter.

27 The following mechanisms of curing reactions were used:

28 – mechanism coded t:f; CnB,Fn,An; for formal kinetic model of A-1 \rightarrow B-2 \rightarrow C-3 \rightarrow D type for
29 DGEBA/EFO/RAMA

30 and

1 degradation. The Avrami–Erofeev and n^{th} reaction models were used (Eqs. 12 and 13). Calculations were
2 performed for several reaction schemes, taking into consideration the formal kinetic reaction model: A-
3 $1 \rightarrow B-2 \rightarrow C$ coded as d:f; (consecutive reactions) and $A-1 \rightarrow B-2 \rightarrow C-3 \rightarrow D$ coded as t:f,f; (consecutive
4 reactions) which show that thermal degradation occurs through successive processes in two and three steps.
5 The choice of the optimal model was made on the basis of the criteria presented above and are shown in
6 Table S2 (See Supporting Information). The activation energies for the first stage of degradation depend on
7 the chemical structure of epoxy oil and have a value of 302 kJ mol^{-1} for DGEBA/EFO/RAMA sample and of
8 323 kJ mol^{-1} for DGEBA/EHO/RAMA sample, while the $\log A/s^{-1}$ factor has similar values for both samples,
9 around 20.

10 Fig. 7_revised

11 Table 2

12 Using the data from Table S2 the TG curves were recalculated. Good overlapping of experimental and
13 calculated data curves may be observed (Fig. 7a).

14 In the case of WECs, the thermal characterization was obtained with same software. Fig. 8a
15 presents the TG and DTA curves of W/DGEBA/EHO/RAMA sample obtained at three heating rates in
16 nitrogen atmosphere. From Fig. 8a, the main degradation parameters (T_{onset} , T_{peak} , T_{10} , T_{20} , T_{50} and W) are
17 extracted and shown in Table 3. Compared with pure polymers, mass losses begin to become significant after
18 $250 \text{ }^{\circ}\text{C}$. The variation of the apparent activation energy and pre-exponential factor obtained with Friedman
19 method is presented in Fig. 8b. From this figure, the variation of the activation energy of the thermal
20 degradation versus the conversion function degree deviates from linearity, indicating that the thermal
21 degradation process takes place in more than two stages. Using the MLR method with An and n^{th} equations,
22 the most probable kinetic model for $A-1 \rightarrow B-2 \rightarrow C$ type, coded as d:f; (consecutive reactions) was
23 obtained. The optimal model version for W/DGEBA/EFO/RAMA sample is encoded as d:f; An , Fn , and for
24 W/DGEBA/EHO/RAMA sample encoded as d:f; An , An . The kinetic and statistical parameters obtained for
25 the above models are shown in Table S3 (See Supporting Information). The simulated TG curves obtained
26 based on the data from Table S3, presented in Fig. 8a, are in a good agreement with experimental data,
27 suggesting that the theoretical models accurately approximate the experimental data. For the sample
28 W/DGEBA/EFO/RAMA, in Table S3 in the second degradation step, the reaction order has a value greater
29 than 1, suggesting that oligomers are generated during the thermal degradation process [55].

30 Table 3

Fig. 8_revised*3.6. Chemical composition of the evolved gases*

In Scheme 3 there is presented the probable mechanism of thermal degradation of the W/DGEBA/EFO/RAMA cured sample. The gaseous products which appear as result of thermal degradation were identified using FT-IR analysis. In Fig. 9a it is shown the 3D FT-IR spectrum and in Fig. 9b the 2D FT-IR spectrum, extracted at the maximum temperature in the Gram Schmidt plot (365 °C) of the evolved gases which appear during thermal degradation of the W/DGEBA/EFO/RAMA cured sample, registered at 10 °C min⁻¹. From the 2D FT-IR spectra (Fig. 9b) it is observed a relatively large number of signals located at 3735 cm⁻¹, 3649 cm⁻¹, 3568 cm⁻¹, 3257 cm⁻¹, 2937 cm⁻¹, 2360 cm⁻¹, 1740 cm⁻¹, 1508 cm⁻¹, 1261 cm⁻¹, 1177 cm⁻¹, 827 cm⁻¹, 669 cm⁻¹ and 538 cm⁻¹. The identification of the signals was made based on literature data [28]. Thus, the signals situated between 3750 and 3500 cm⁻¹ can be attributed to OH groups from water vapors or alcohols which resulted from the thermal degradation of the ester and secondary hydroxyl groups. On the other hand, the signal located at 3257 cm⁻¹ can be assigned to secondary alcohols and phenols which resulted from degradation of lignin and epoxy resin moieties [56]. The signals assigned to CH, CH₂ and CH₃ groups specific to aliphatic and aromatic hydrocarbon residues are located between 3101 cm⁻¹ and 2800 cm⁻¹, as well as 1652 cm⁻¹ and 1461 cm⁻¹. These groups result from oil and DGEBA moieties degradation. The strong signal from 2360 cm⁻¹ can be attributed to carbon dioxide from the degradation of ester groups. Hemicelluloses and cellulose from wood are mainly composed of D-glucopyranose rings, linked together by 1-4-glycosidic bonds. By cleaving these linkages, D-glucopyranose rings appear which, by recombination, lead to levoglucosan formation. With temperature rise and in the presence of water levoglucosan is transformed into carboxylic derivatives. The signals between 1860 to 1650 cm⁻¹ with a maximum located at 1740 cm⁻¹ can be attributed to acids, ketones, aldehydes and esters moieties, resulted from the degradation of wood, RAMA and oil. The signals between 1650 cm⁻¹ and 1500 cm⁻¹ corroborate with the signal situated at 3014 cm⁻¹ and can be assigned to aromatic rings from benzene, toluene, phenol, α -methyl styrene, substances that originate from the degradation of DGEBA, RAMA and lignin moieties. The signals situated at 1261 and 1177 cm⁻¹ can be assigned to ether C-O-C group.

Scheme 3**Fig. 9_revised**

1 3.7. Resistance to fungi

2 After 10 weeks of exposure to fungi, both the untreated and the polymer treated samples are
3 covered with fungi colonies, more the blank sample and lesser the polymer coated sample (Fig. 10). After
4 washing and conditioning, the untreated wood sample loses up to 19.4 wt. % depending on the type of
5 fungus, while for the polymer coated wood samples, their weight loss is no more than 2 wt. % (Table 4). It
6 can be seen that the strongest attack is due to *A. brasiliensis* and *C. cladosporioides* for the untreated samples.
7 The polymer covered samples were more attacked by *A. brasiliensis* and *P. chrysogenum*. In the case of
8 pure polymers, all fungi developed very small surface colonies, suggesting that by depositing a thicker layer
9 of polymer may generate an important wood protection effect. Also, in Table 4 there are presented the
10 humidity values after 10 weeks of incubation to fungi exposure. It can be seen that the amount of water
11 absorbed depends on the type of polymer and fungus, the smallest amount being absorbed by
12 W/DGEBA/EHO/RAMA.

13 **Fig. 10_revised**

14 **Table 4**

15 16 3.8. WAXD characterization

17 Figure 11 shows the WAXD curves of the blank and covered samples registered before and after the
18 fungi decay tests. In general, for pure cellulose, four crystalline peaks (101,10 $\bar{1}$, 002 and 040) have been
19 identified [57]. In our case, the major diffraction peak specific to the crystalline cellulose is located in the
20 range 22.2 – 22.45°, while the diffraction peak specific for the amorphous phases is situated between 18.6
21 and 18.8° [58, 59]. The fungi preferentially attack the amorphous regions of cellulose, since these regions are
22 less crosslinked in comparison to the crystalline region [60, 61]. These regions may also be highlighted by
23 X-ray analysis. From Table 5 one may observe that the crystallinity index decreases as a result of fungi
24 attack, the lower values being encountered for the unprotected wood samples in comparison with protected
25 samples. This may be due to the blocking by the polymer film of the fungi access to wood.

26 **Fig. 11_revised**

27 **Table 5**

28 29 3.9. SEM and EDAX measurements

1 As an exemplification, the SEM images obtained in radial section with 1 mm thickness for wood
2 and W/DGEBA/EFO/RAMA sample, decayed in the presence of *C. cladosporioides*, *P. chrysogenum* and *A.*
3 *brasiliensis* are presented in Figure 12. From these images, it can be seen that the raw wood sample masses
4 have developed a large number of fungus hives inside, while the samples protected with
5 DGEBA/EFO/RAMA were not affected. The results are in good agreement with literature data [62]. Using
6 the EDAX analysis it was possible to evaluate the changes in the elemental chemical composition of the
7 surface for raw wood and treated samples after biological decay. Table 6 shows the chemical composition of
8 the sample surfaces for wood and the coated samples. It can be noted that for all the samples there is a
9 significant increase in the percentage of nitrogen as a result of fungal growth. Also, an increase in the
10 percentage of oxygen for the unprotected samples can be observed, which may be due to the appearance of
11 some carboxyl derivatives occurred during cellulose biodegradation. In the case of wood, there appears a
12 significant decrease in the carbon percentage due to the consumption of wood components. The decrease of
13 microelements content may be attributed to the mass increase of the organic product due to the coating
14 process.

15 **Fig. 12**

16 **Table 6**

18 **Conclusions**

19 New wood composite materials based on synthetic and natural epoxy derivatives have been
20 obtained and tested in terms of the influence on the thermal stability and fungi resistance. The wood
21 composites were obtained by immersion of the wood samples in a solution of DGEBA/EO/RAMA in
22 acetone and thermally cured. The kinetic parameters of the curing reaction of DGEBA/EO/RAMA and the
23 thermal characterization of crosslinked DGEBA/EO/RAMA and WECs were obtained using DSC and TG
24 methods. The most probable mechanisms of the curing and thermal degradation processes occurred in two or
25 three steps. The decay resistance of wood, DGEBA/EO/RAMA polymers and WECs was obtained by
26 exposure to three fungi (*A. brasiliensis*, *P. chrysogenum* and *C. cladosporioides*) and evaluated with the aid
27 of WAXD and SEM/EDAX methods. Under fungi action, the crystallinity index decreased, the largest drop
28 being registered under the attack of *A. brasiliensis*. Also, the mass loss was higher after the attack of *A.*
29 *brasiliensis*, reaching 19.4 wt. % for wood, while for the WECs this loss was only 2 wt. %. The polymer

1 coatings blocked the access of the fungi in the sample masses, thus inducing a superior resistance to their
2 attack.

3

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9 SEM/EDAX measurements.

10

11 **Conflict of interest**

12 The authors declare no conflict of interest.

13

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29

30

1 **Schemes and Figures captions**

2 **Scheme 1.** The synthesis of epoxy oil.

3 **Fig.1.** Variation of the swelling coefficient with the solubility parameter for: (●) DGEBA/EFO/RAMA and
4 (○) DGEBA/EHO/RAMA.

5 **Scheme 2.** The possible mechanism of the thermal curing of the DGEBA/EO/RAMA sample.

6 **Fig.2.** FT–IR spectra of flax oil (a) and epoxidized flax oil (b).

7 **Fig.3.** ¹H–NMR spectra of flax oil (a) and epoxidized flax oil (b).

8 **Fig.4.** FT–IR spectra of W (a), crosslinked DGEBA/EFO/RAMA (b) and WEC (c).

9 **Fig.5** DSC curves for DGEBA/EFO/RAMA (a) and DGEBA/EHO/RAMA (b) at: (■) 5 °C min⁻¹, at (●) 10
10 °C min⁻¹ and at (▲) 15 °C min⁻¹ (symbols represent experimental data and lines represent simulated data).

11 **Fig.6.** Dependence of the (■) activation energy and the (●) pre–exponential factor versus the conversion
12 degree for the curing of: DGEBA/EFO/RAMA (a) and DGEBA/EHO/RAMA (b) calculated with Friedman
13 method.

14 **Fig.7.** TG and DTG curves for: DGEBA/EFO/RAMA at: (■) 10 °C min⁻¹, at (●) 20 °C min⁻¹ and at (▲) 30
15 °C min⁻¹ (symbols represent experimental data and lines represent simulated data) (a). Dependence of the
16 (■) activation energy and the (●) pre–exponential factor versus the conversion degree for
17 DGEBA/EFO/RAMA calculated with Friedman method (b).

18 **Fig.8.** TG and DTG curves for: DGEBA/EHO/RAMA at: (■) 10 °C min⁻¹, at (●) 20 °C min⁻¹ and at (▲) 30
19 °C min⁻¹ (symbols represent experimental data and lines represent simulated data) (a). Dependence of the
20 (■) activation energy and the (●) pre–exponential factor versus the conversion degree for
21 DGEBA/EHO/RAMA calculated with Friedman method (b).

22 **Scheme 3.** The possible mechanism of the thermal degradation of the cured DGEBA/EO/RAMA sample.

23 **Fig.9.** 3D FT–IR and 2D FT–IR spectra, of the evolved gases during thermal degradation of the
24 W/DGEBA/EFO/RAMA cured sample, registered at 10 °C min⁻¹ heating rate.

25 **Fig.10.** Blank and the polymer-treated sample exposed to fungi: (a) W, W/DGEBA/EHO/RAMA and
26 DGEBA/EHO/RAMA exposed to *A. brasiliensis*; (b) W, W/DGEBA/EHO/RAMA and
27 DGEBA/EHO/RAMA exposed to *C. cladosporioides*; (c) W, W/DGEBA/EHO/RAMA and
28 DGEBA/EHO/RAMA exposed to *P. chrysogenum*; (d) W, W/DGEBA/EFO/RAMA and
29 DGEBA/EFO/RAMA exposed to *A. brasiliensis* (e) W, W/DGEBA/EFO/RAMA and DGEBA/EFO/RAMA

1 exposed to *C. cladosporioides*; (f) W, W/DGEBA/EFO/RAMA and DGEBA/EFO/RAMA exposed to *P.*
2 *chrysogenum*

3 **Fig.11.** (a) WAXD curves for: W (1); W decayed with *A. brasiliensis* (2); W decayed with *C.*
4 *cladosporioides* (3); W decayed with *P. chrysogenum* (4); (b) WAXD curves for: DGEBA/EHO/RAMA
5 decayed with *C. cladosporioides* (1); DGEBA/EHO/RAMA undecayed (2); DGEBA/EHO/RAMA decayed
6 with *A. brasiliensis* (3); and DGEBA/EHO/RAMA decayed with *P. chrysogenum* (4); (c) WAXD curves for:
7 DGEBA/EFO/RAMA decayed with *P. chrysogenum* (1); DGEBA/EFO/RAMA decayed with *C.*
8 *cladosporioides* (2); DGEBA/EFO/RAMA decayed with *A. brasiliensis* (3) and DGEBA/EFO/RAMA
9 undecayed (4).

10 **Fig.12.** SEM micrographs of the cross-section of W decayed with: *C. cladosporioides* (a), *P. chrysogenum*
11 (b) and *A. brasiliensis* (c); W/DGEBA/EFO/RAMA decayed with: *C. cladosporioides* (d), *P. chrysogenum*
12 (e) and *A. brasiliensis* (f).

13

Table 1. Solvents and their corresponding solubility parameters (δ)

Number	Solvent	δ (cal cm ⁻³) ^{0.5}
1	n-Hexane	7.24
2	Carbon tetrachloride	8.05
3	Cyclohexane	8.18
4	Xilene	8.85
5	Toluene	8.91
6	Ethyl acetate	9.1
7	Chloroform	9.21
8	Tetrahydrofurane	9.52
9	Acetone	9.77
10	Cyclohexanone	9.88
11	Methylene chloride	9.93
12	Dioxane	10
13	Acetic acid	10.1
14	Dimethyl acetamide	11.0
15	Isopropyl alcohol	11.6
16	Dimethyl formamide	12.14
17	Dimethyl sulfoxide	12.93
18	Methanol	14.28
19	Diethylene glycol	14.6
20	Ethylene glycol	16.3
21	Glycerol	21.1
22	Water	23.5

Table 2.

Thermal degradation parameters obtained by TG analysis of cured samples at three heating rates.

Sample	Heating rate	T _{onset}	T _{peak}	W ₆₀₀	T ₁₀	T ₂₀	T ₅₀
	°C·min ⁻¹	°C	°C	%	°C	°C	°C
DGEBA/EFO/RAMA	10	380	422	1.2	259	374	413
	20	380	427	2.1	325	374	418
	30	378	430	5.2	331	375	425
DGEBA/EHO/RAMA	10	383	419	1.1	339	380	419
	20	383	429	2.2	355	387	429
	30	377	436	5.4	363	390	499

T_{onset} – the temperature at which the thermal degradation start;T_{peak} – the temperature at which the degradation rate is maximum;T₁₀, T₂₀, T₅₀ - the temperatures corresponding to 10 wt.%, 20 wt.%, 50 wt.% mass losses;W₆₀₀- residual mass at 600 °C

Table 3.

The thermal parameters obtained by TG analysis of cured W/DGEBA/EO/RAMA samples at three heating rates

Sample	Heating rate	T _{onset}	T _{peak}	W ₆₀₀	T ₁₀	T ₂₀	T ₅₀
	°C·min ⁻¹	°C	°C	%	°C	°C	°C
W/DGEBA/EFO/RAMA	10	319	356	17.4	286	320	361
	20	322	363	27.8	286	326	370
	30	323	365	20.9	285	328	378
W/DGEBA/EHO/RAMA	10	321	354	20.5	296	323	362
	20	329	364	22.9	302	330	380
	30	320	371	20.2	298	330	381

T_{onset} – the temperature at which the thermal degradation starts;

T_{peak} – the temperature at which the degradation rate is maximum;

T₁₀, T₂₀, T₅₀ - the temperatures corresponding to 10 wt.%, 20 wt.%, 50 wt.% mass losses;

W₆₀₀ - residual mass at 600 °C

Table 4. Some characteristic of the samples, registered after a 10 week exposure against *A. brasiliensis*, *P. chrysogenum* and *C. cladosporioides* strains.

Sample	mean weight loss ^a			humidity ^a		
		WL (%)			H (%)	
	<i>A. brasiliensis</i>	<i>P. chrysogenum</i>	<i>C. cladosporioides</i>	<i>A. brasiliensis</i>	<i>P. chrysogenum</i>	<i>C. cladosporioides</i>
W	19.4	4.47	6.37	50.4	50.5	33.5
W/DGEBA/EHO/RAMA	1.82	1.38	0.43	6.97	22.6	26.7
W/DGEBA/EFO/RAMA	1.89	1.39	1.22	38.5	36.9	16.7

a) Means were for 4 samples

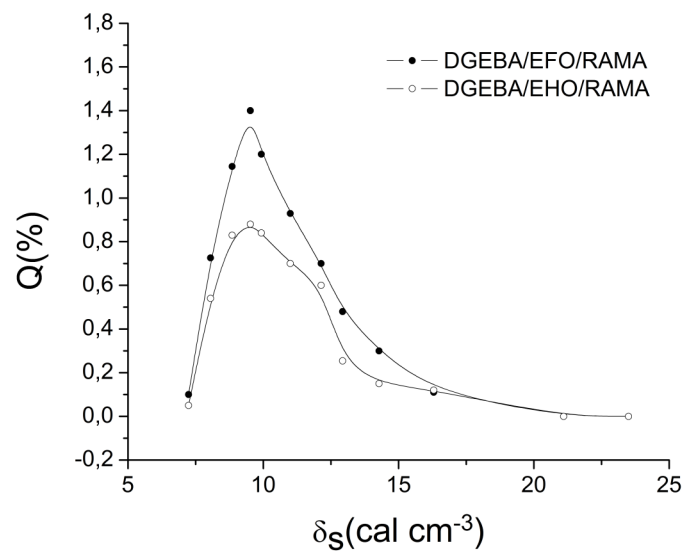
Table 5. Crystalline index

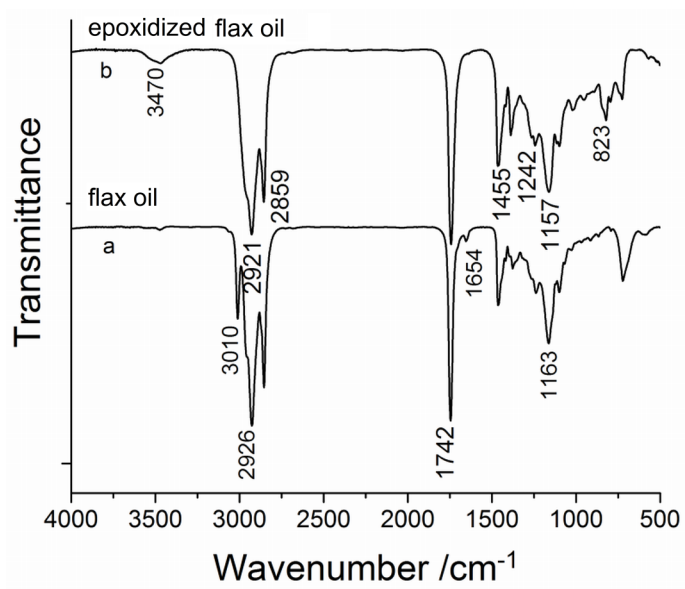
Sample	Undecayed blank	<i>A. brasiliensis</i> decayed	<i>P. chrysogenum</i> decayed	<i>C. cladosporioides</i> decayed
W	74.0	67.58	70.04	70.1
W/DGEBA/EHO/RAMA	67.4	63.2	64.4	64.13
W/DGEBA/EFO/RAMA	69.2	63.6	67.7	67.0

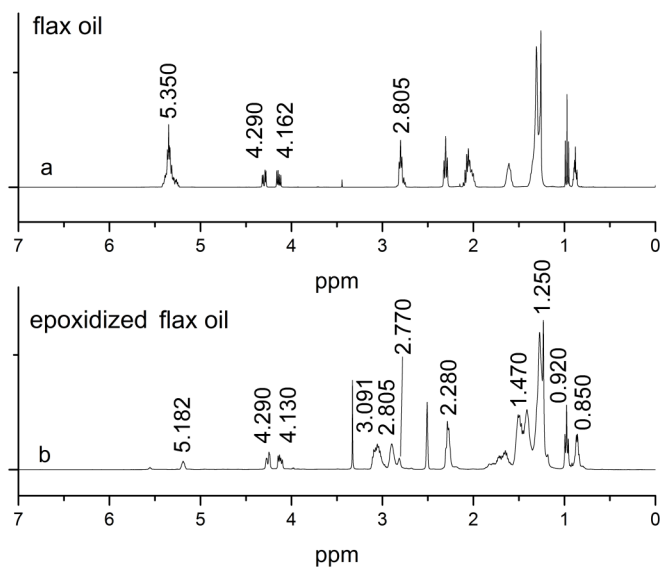
Means were for 4 samples

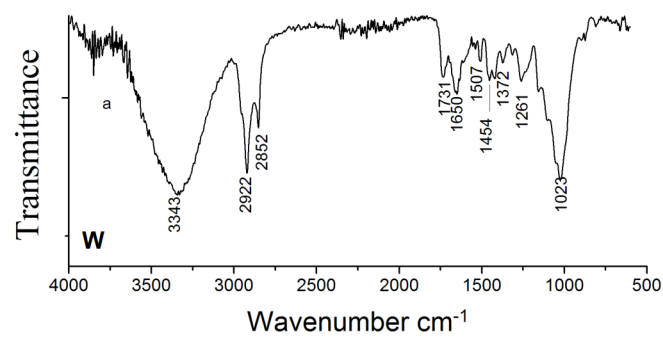
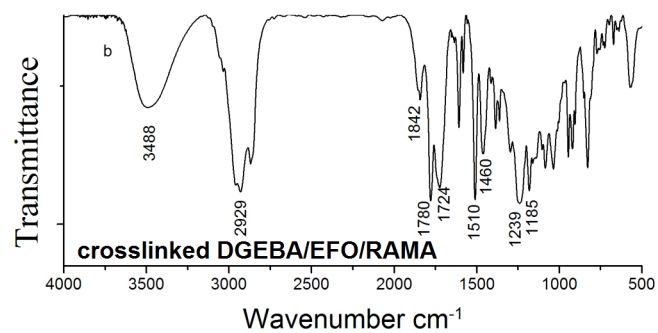
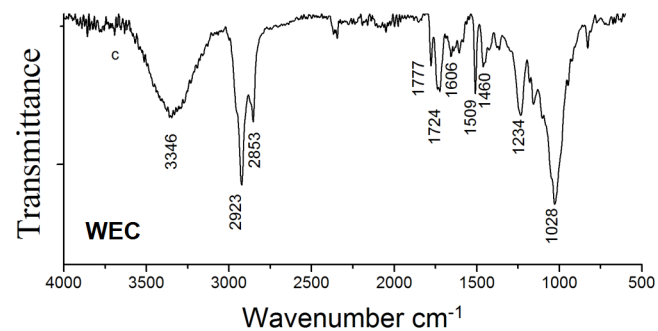
Table 6. SEM/EDAX composition of wood and W/DGEBA/EO/RAMA samples

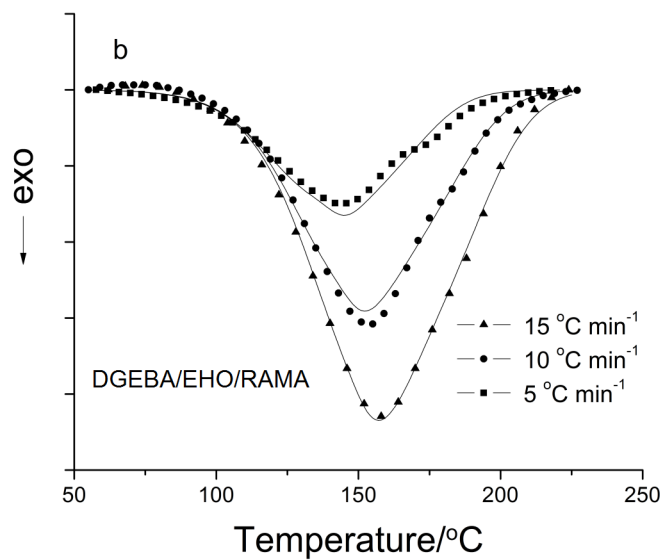
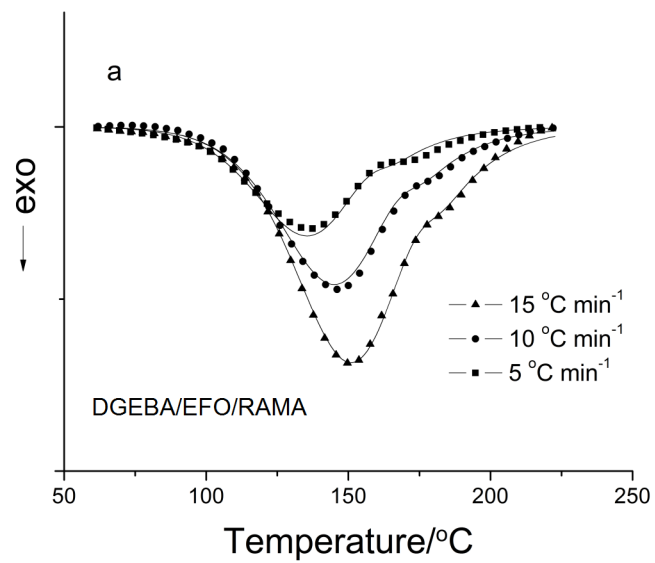
Element	Sample									
	W			W/DGEBA/EHO/RAMA			W/DGEBA/EFO/RAMA			
	undecayed	decayed with	decayed with	decayed with	decayed with	decayed with	decayed with	decayed with	decayed with	decayed with
	<i>C. cladosporioides</i>	<i>P. chrysogenum</i>	<i>A. brasiliensis</i>	<i>C. cladosporioides</i>	<i>P. chrysogenum</i>	<i>A. brasiliensis</i>	<i>C. cladosporioides</i>	<i>P. chrysogenum</i>	<i>A. brasiliensis</i>	
C	66.2	58.42	58.45	55.85	71.68	70.52	71.94	68.65	58.30	69.08
N	1.66	4.60	3.55	4.51	4.62	6.10	5.45	3.77	5.31	4.72
O	30.11	35.23	36.82	36.4	23.63	25.2	21.20	33.22	35.15	32.63
Fe	0.62	0.63	0.30	0.61	0.33	0.39	0.25	0.47	0.23	0.63
Na	0.24	0.23	0.24	0.45	0.10	0.15	0.11	0.16	0.13	0.11
Mg	0.04	0.04	0.05	0.07	0.09	0.06	0.20	0.08	0.10	0.08
Al	0.18	0.11	0.06	0.06	0.08	0.12	0.29	0.11	0.12	0.12
Si	0.25	0.14	0.16	0.21	0.12	0.10	0.11	0.10	0.13	0.11
Cl	0.17	0.15	0.14	0.02	0.18	0.15	0.12	0.09	0.12	0.19
K	0.16	0.20	0.17	0.15	0.17	0.08	0.89	0.07	0.14	0.13
Ca	0.16	0.16	0.13	0.14	0.11	0.10	0.19	0.15	0.12	0.11

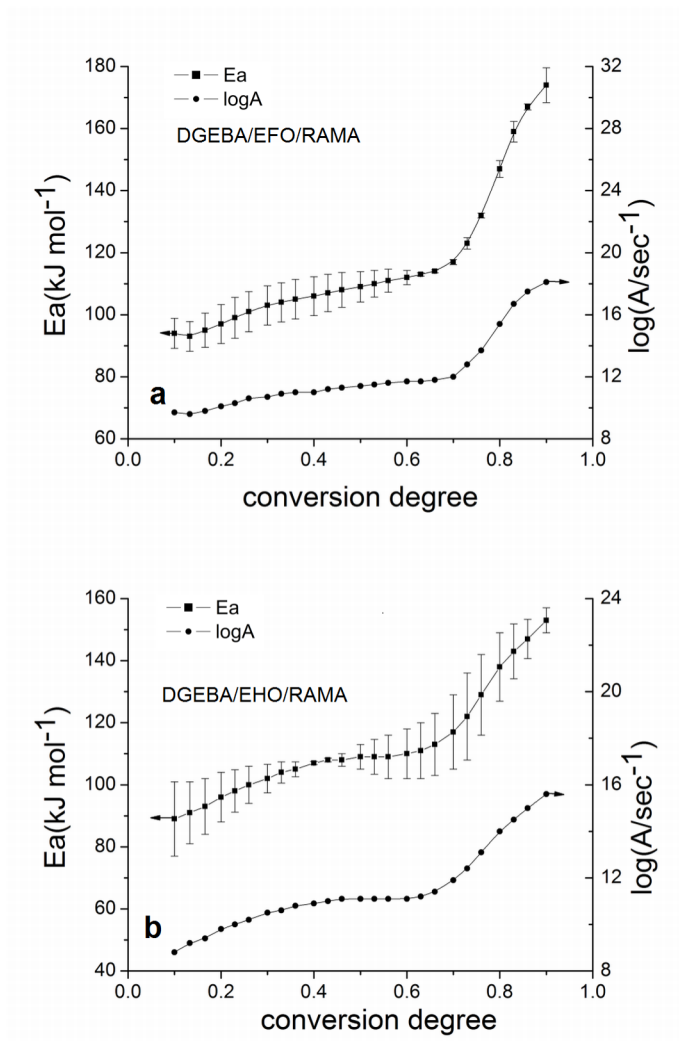


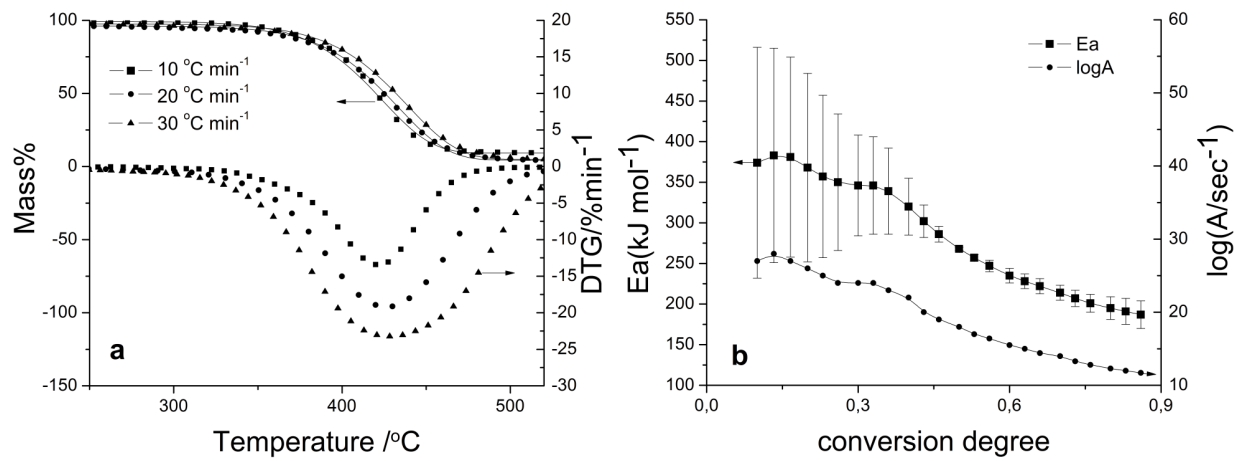


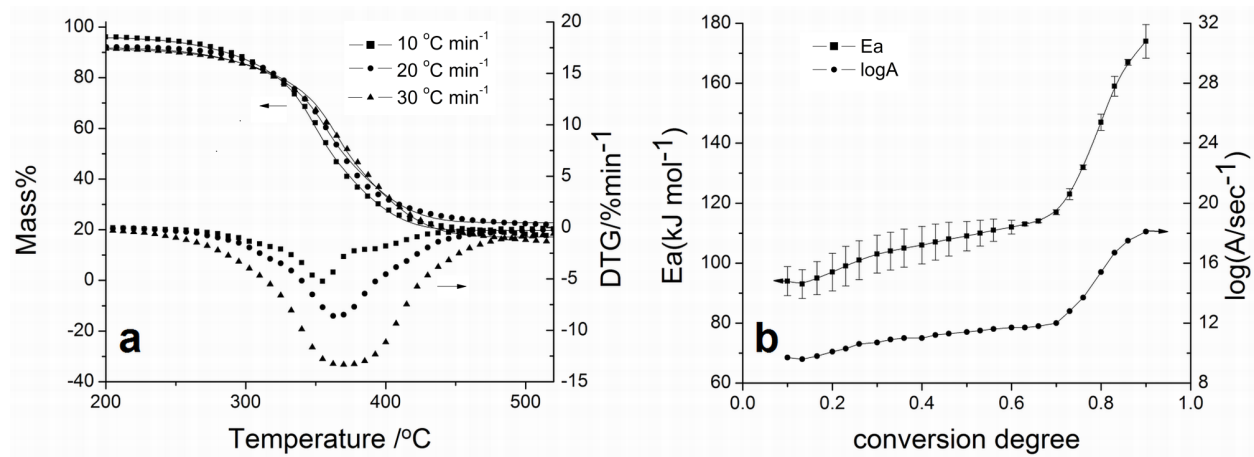


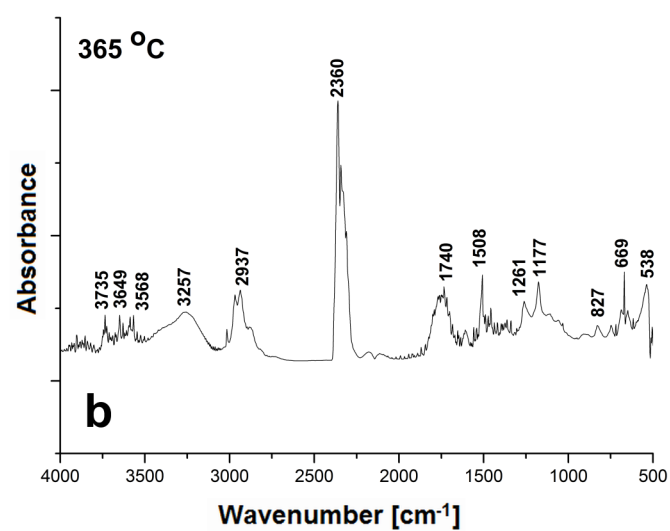
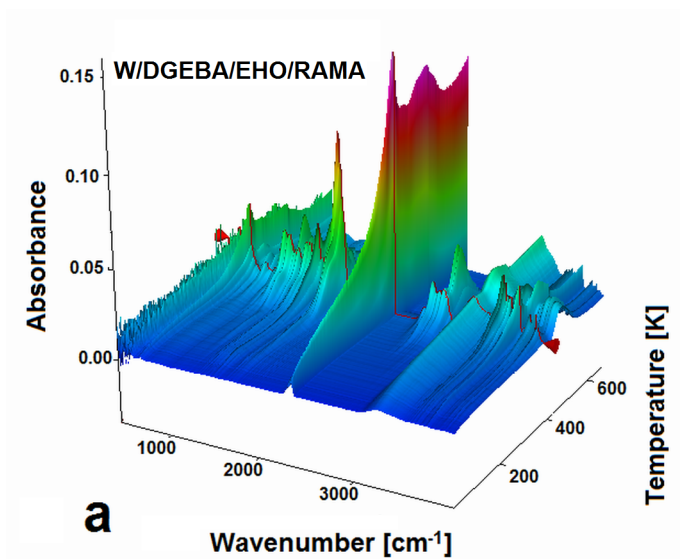


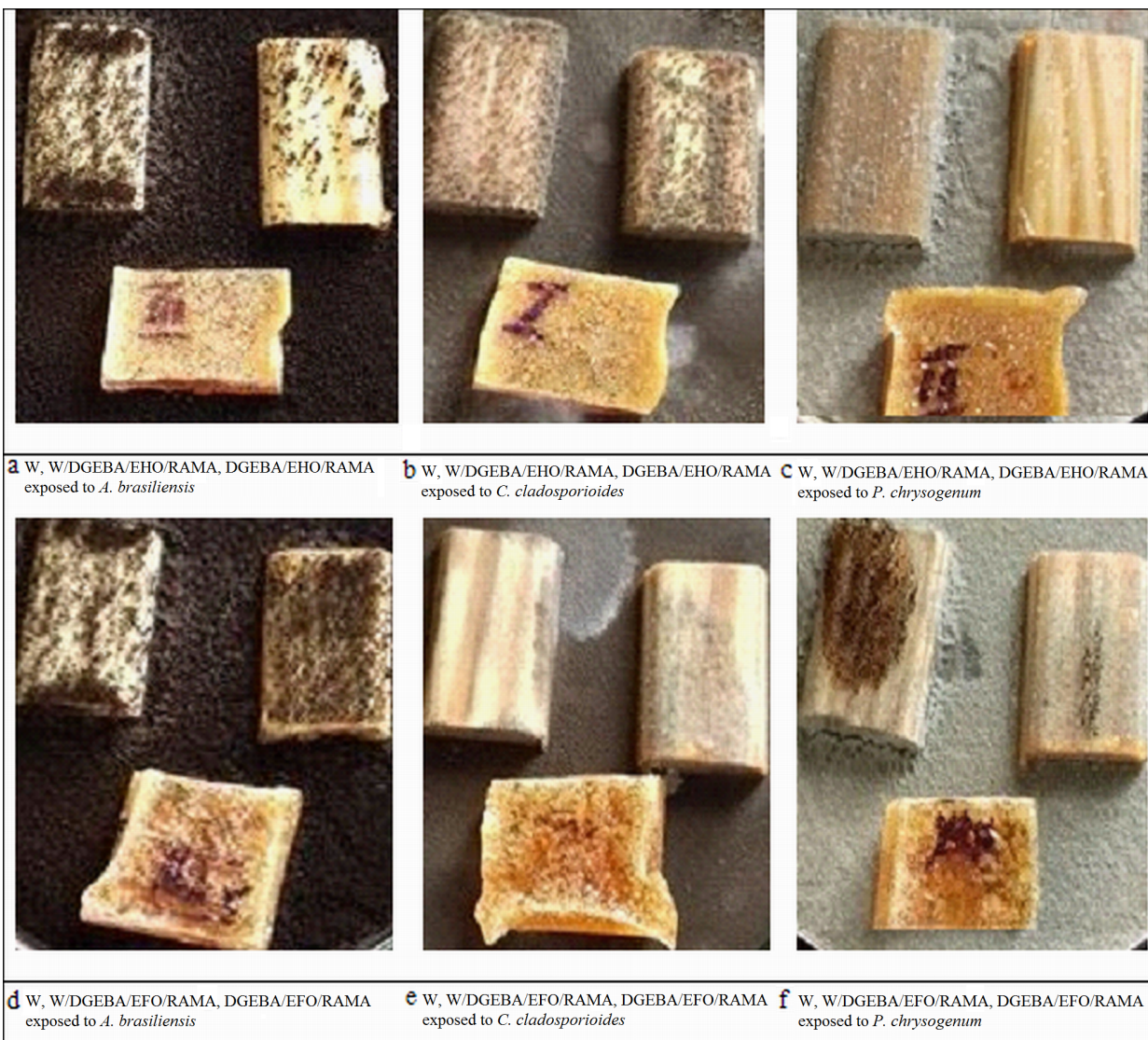




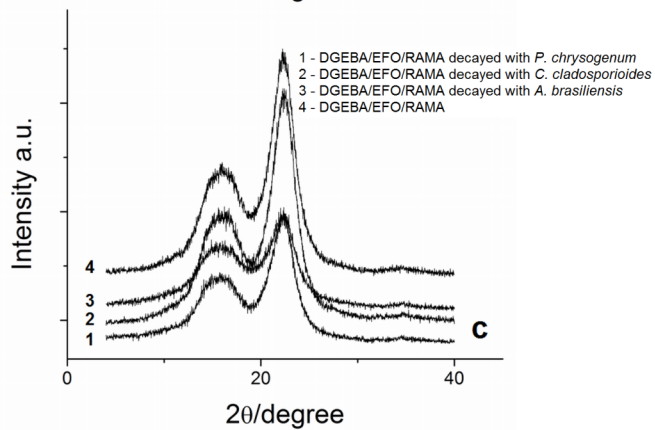
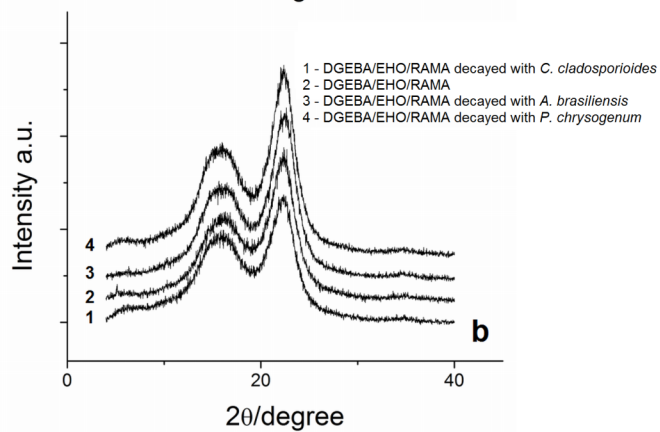
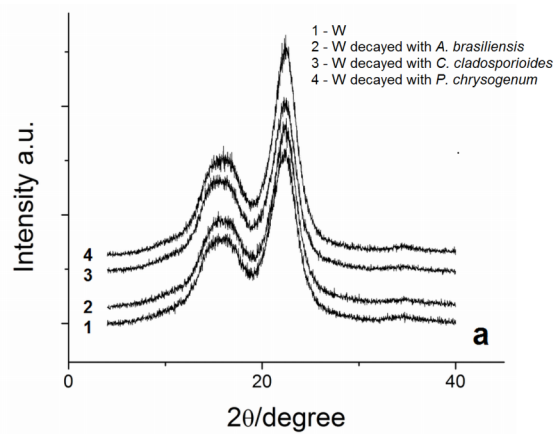


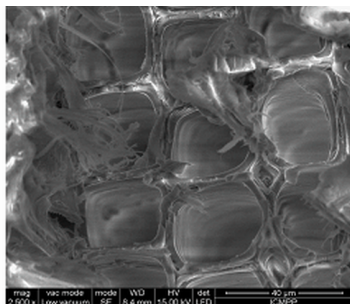
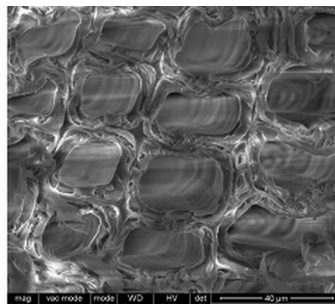
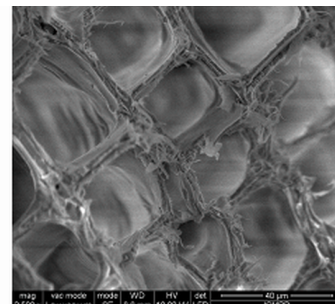
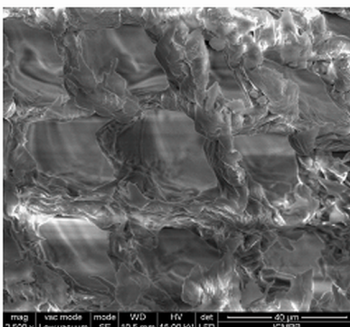
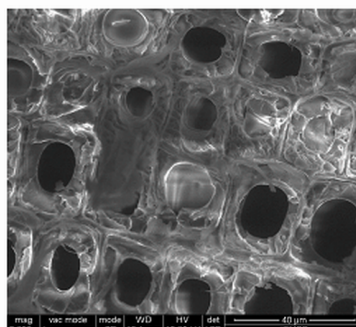
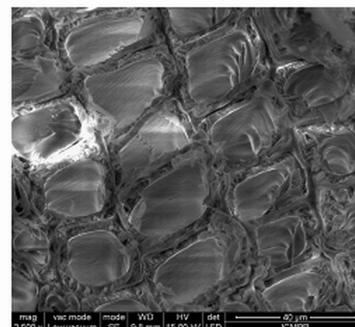




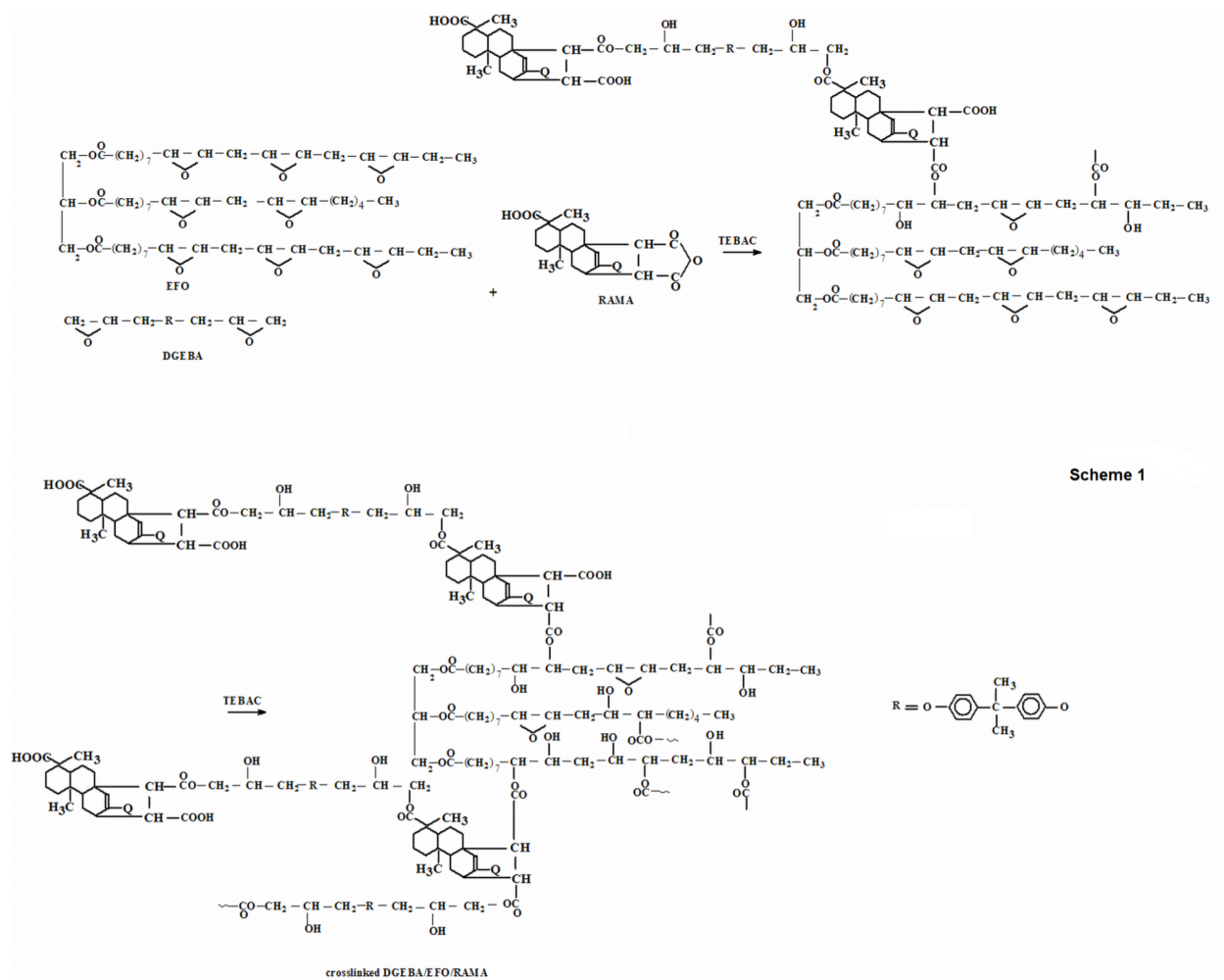


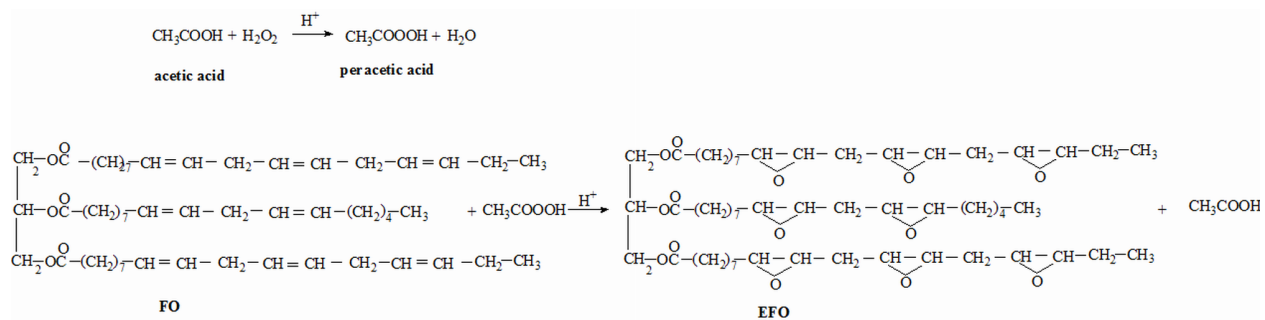
ACCEPTED



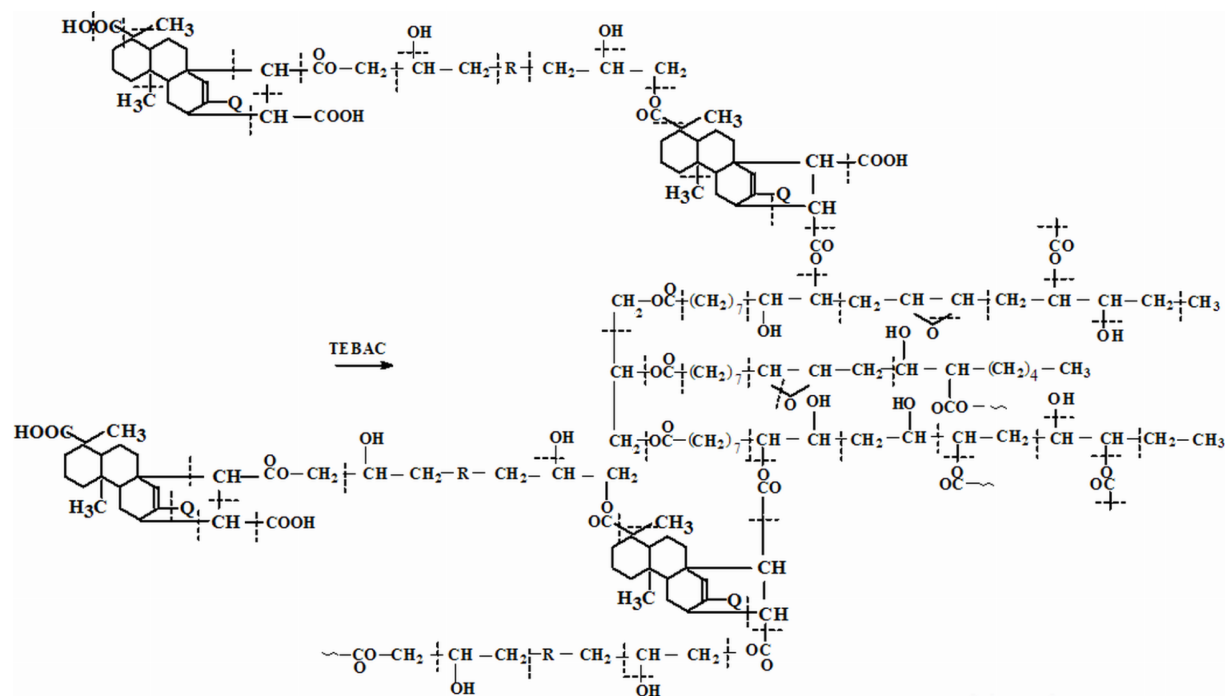
W decayed with *Cladosporium cladosporioides* (a)W decayed with *Penicillium chrysogenum* (b)W decayed with *Aspergillus brasiliensis* (c)W/DGEBA/EFO/RAMA decayed with *Cladosporium cladosporioides* (d)W/DGEBA/EFO/RAMA decayed with *Penicillium chrysogenum* (e)W/DGEBA/EFO/RAMA decayed with *Aspergillus brasiliensis* (f)

ACCEPTED MANUSCRIPT





Scheme 2



Scheme 3

- New synthetic and natural epoxy derivatives as wood coatings were obtained.
- The epoxy derivatives were thermally crosslinked.
- The influence of coating on wood thermal stability and fungi resistance was made.

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